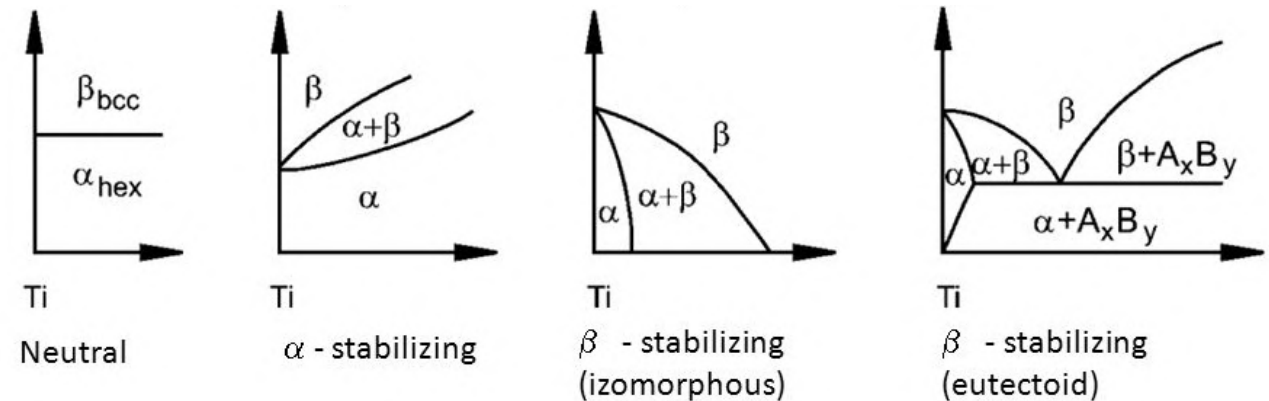


Course goal: Describe phase transformation in Ti and Zr alloys during heating and cooling at different rates and explain how this can change the properties of the material.

Learning outcomes:

- *Describe the effect of different alloying additions on the phase structure of Ti (and Zr) alloys*



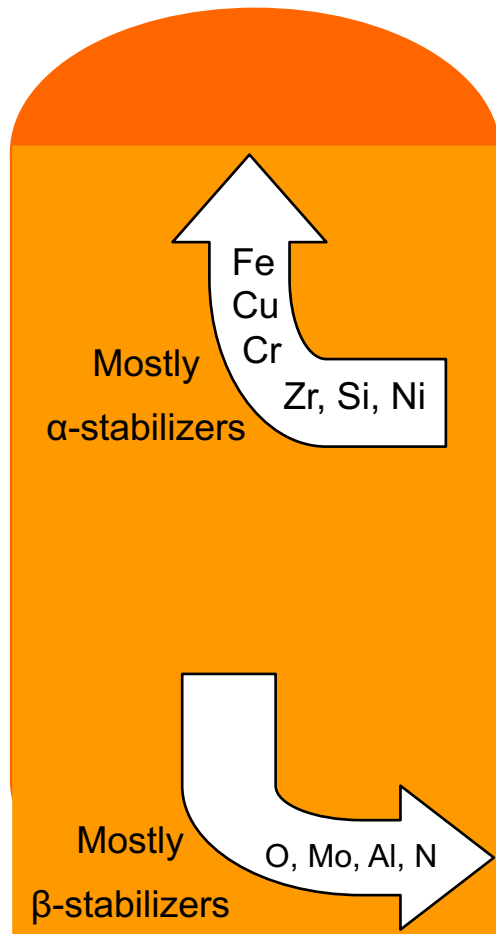
α -stabilising increase β -transus

β -stabilising decrease β -transus

Isomorphous – completely soluble in solid solution

Eutectoid – intermetallic particles are created

Melting – Element Segregation Trends



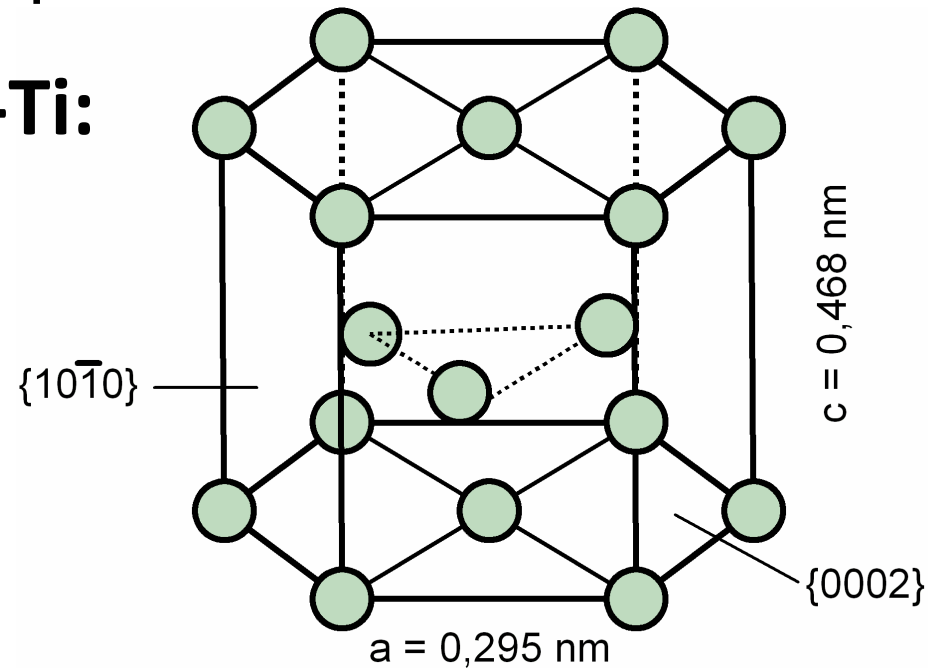
- Subsequent heat treatment / forging will not reduce any macro segregation → ***re-melting***.
- For ‘difficult’ alloys product chemistry and structure is more reliable than ingot analysis.
- For good melt practice oxygen ‘pick up’ is small (< 150 ppm).

When used in aeroengines, melting defects can cause component failure – very rare today.

Allotropic transformation of pure Ti (and Zr)

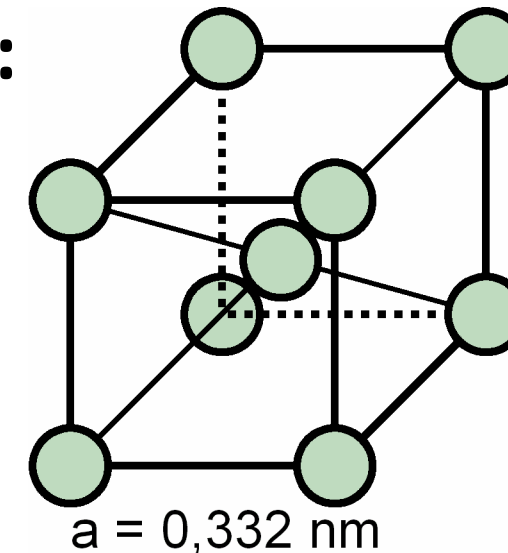
$T < 880^{\circ}\text{C}$

hcp
 $\alpha\text{-Ti}$:



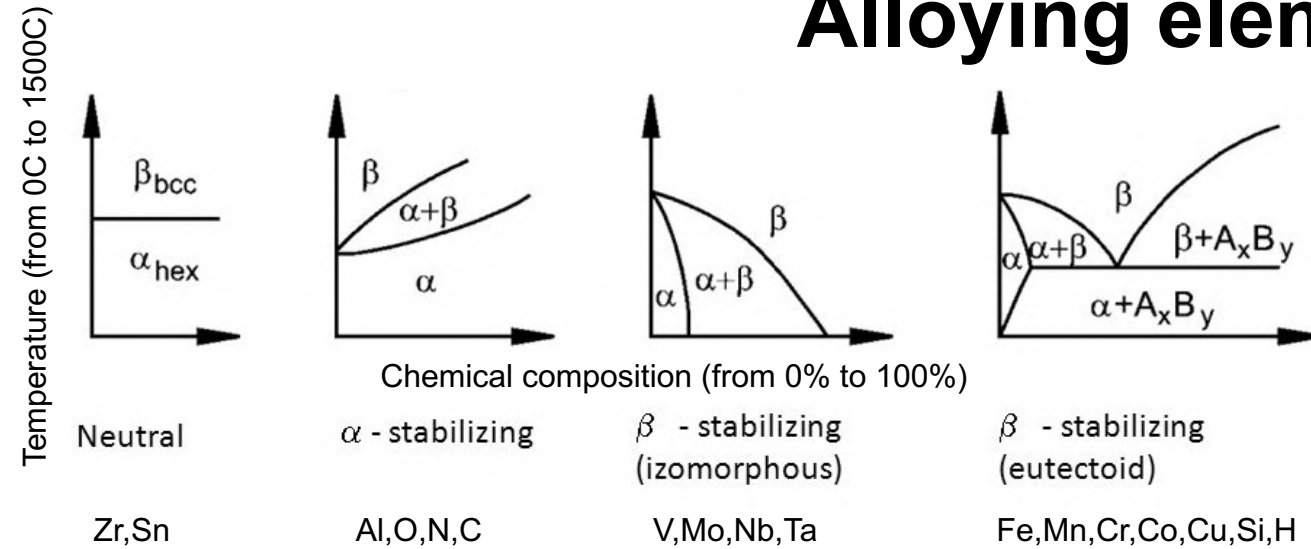
$T > 880^{\circ}\text{C}$

bcc
 $\beta\text{-Ti}$:



Live lecture – Which crystal structure has higher density?

Alloying elements in Ti



α -stabilising increase β -transus
 β -stabilising decrease β -transus
 Isomorphous – completely soluble in solid solution
 Eutectoid – intermetallic particles are created

α -stabiliser

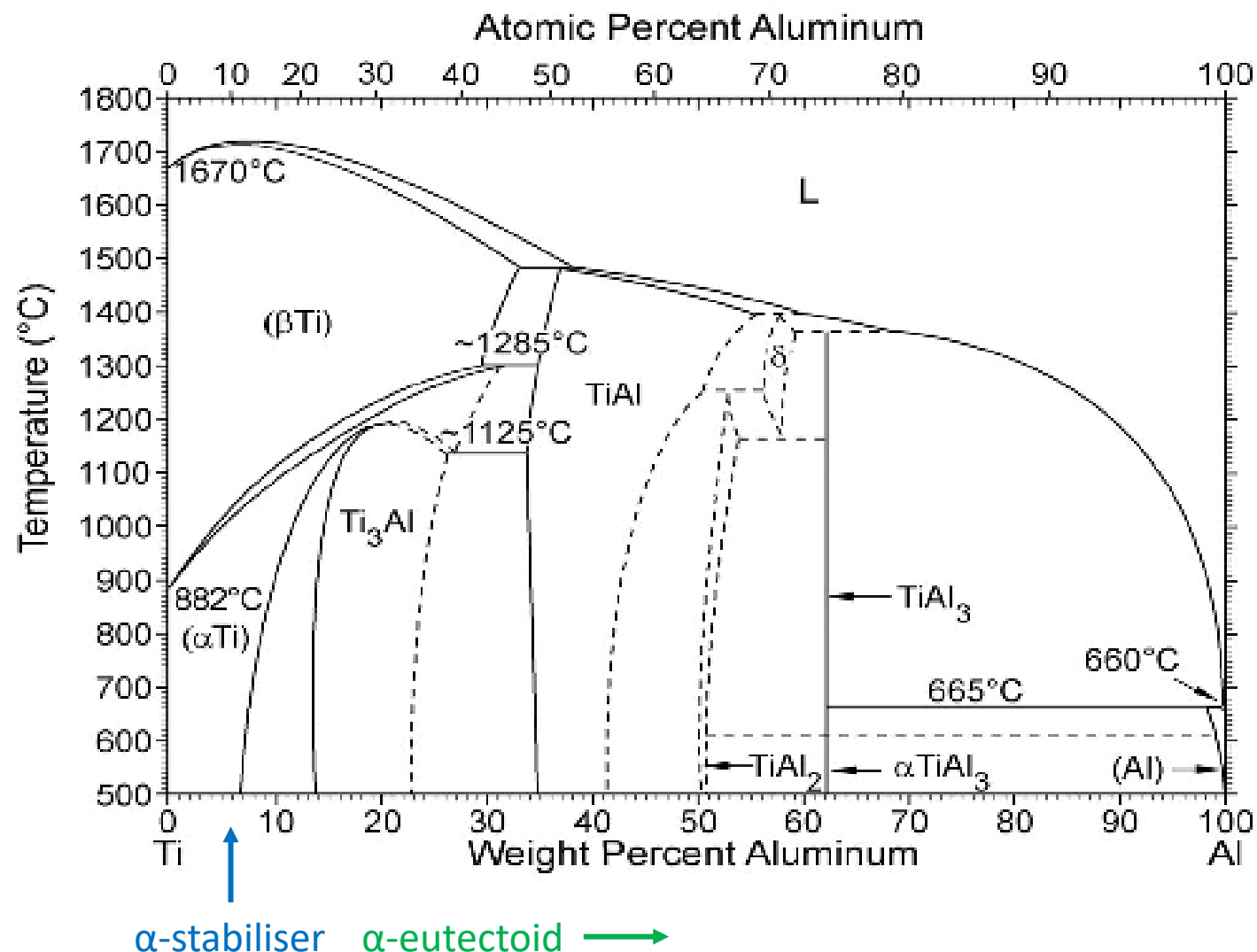
- Stabilise the α -phase at room temperature.
- Eg. Aluminium, Oxygen etc.

β -stabiliser

- Stabilise the β -phase at room temperature.
- Eg. Vanadium, Niobium, Molybdenum, etc.

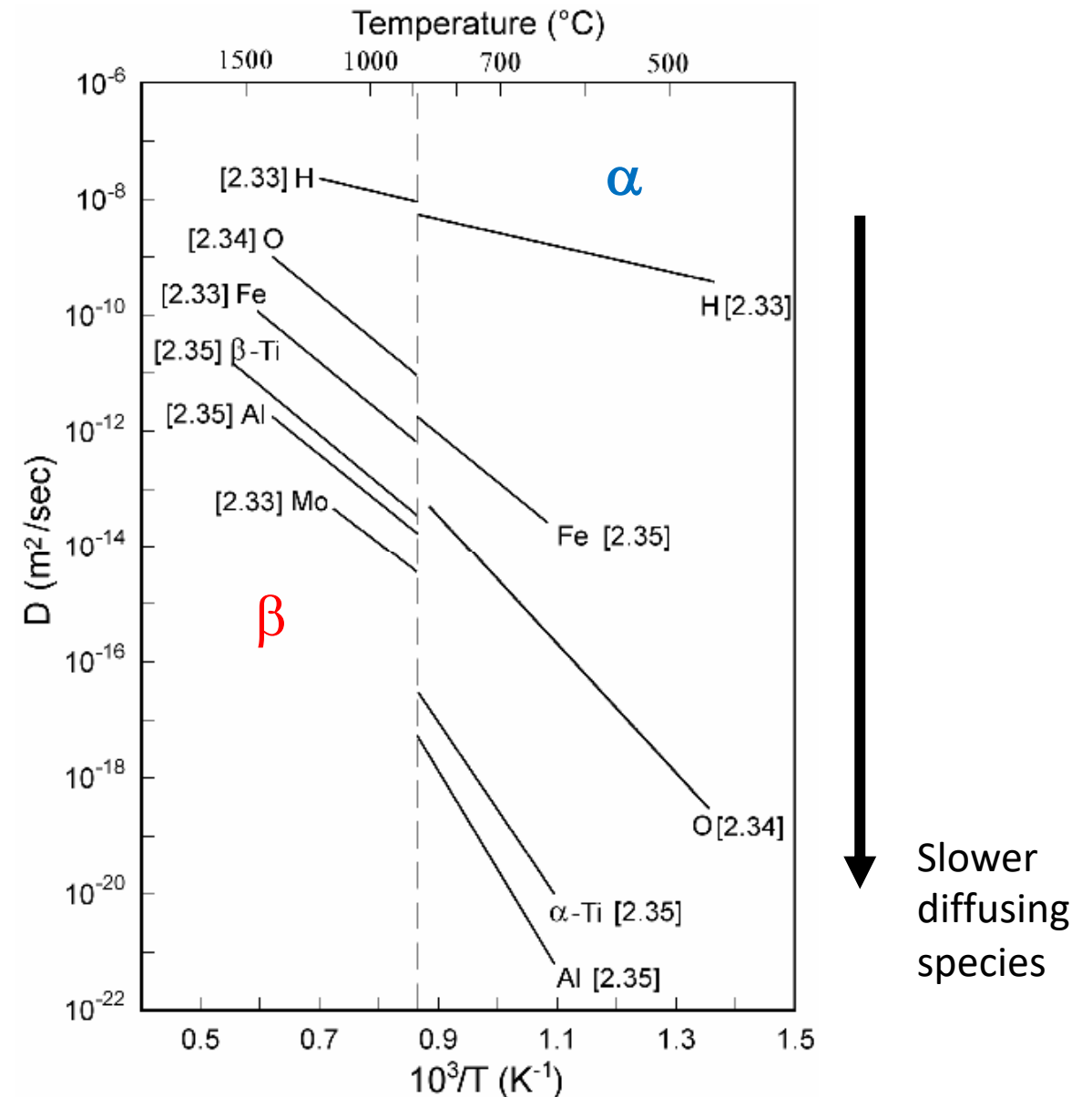
Ti-Al Phase Diagram

- **Thermodynamics** → calculation of system energy at equilibrium → phase diagram
- How fast changes happen? → **Kinetics**
 - **Cooling rate**
 - **Element diffusion**



Diffusion in Ti / Zr

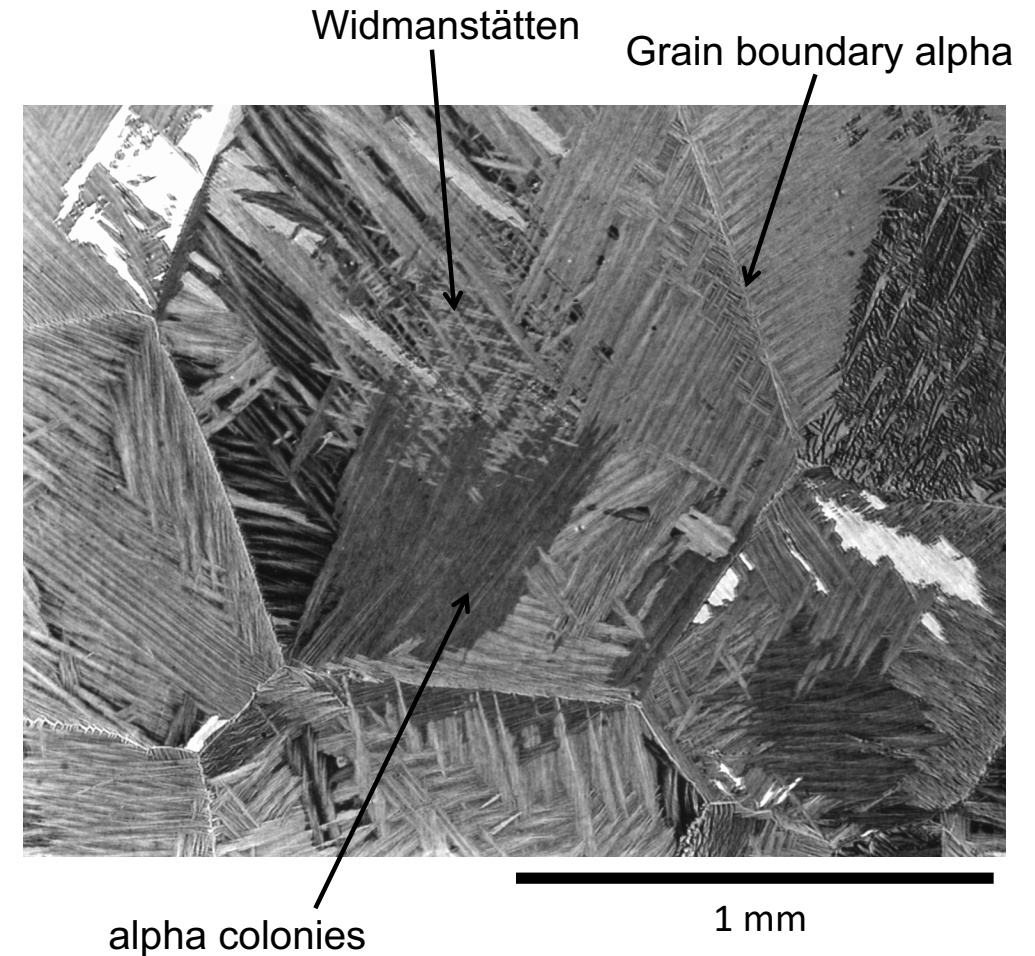
- Arrhenius diagram of titanium self-diffusion and various alloying elements
- Zirconium displays very similar behaviour (*although Nb is a much slower diffusing species in Zr*).



Course goal: Describe phase transformation in Ti and Zr alloys during heating and cooling at different rates and explain how this can change the properties of the material.

Learning outcomes:

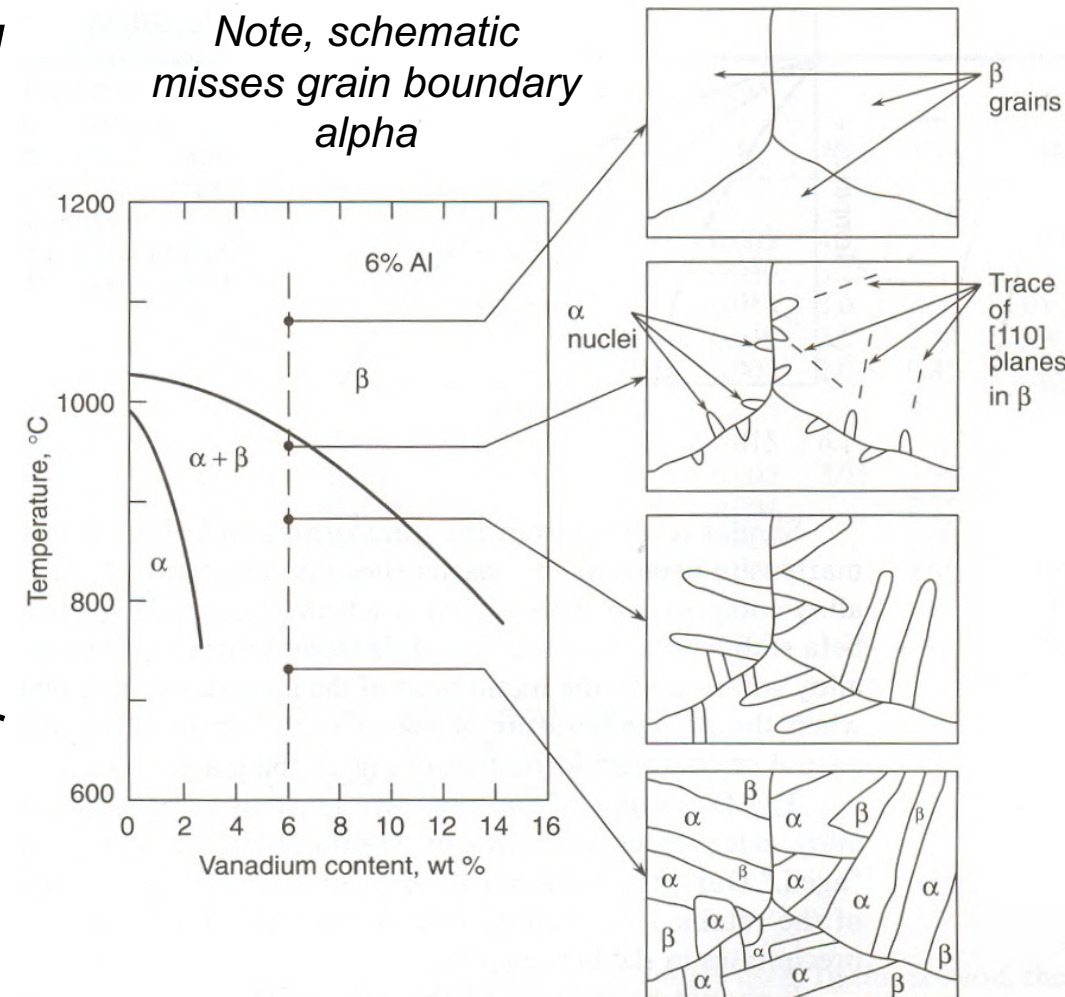
- Describe the effect of different alloying additions on the phase structure of Ti (and Zr) alloys.
- ***Describe phase transformation and microstructural changes during heating and cooling of Ti and Zr alloys at different rates.***



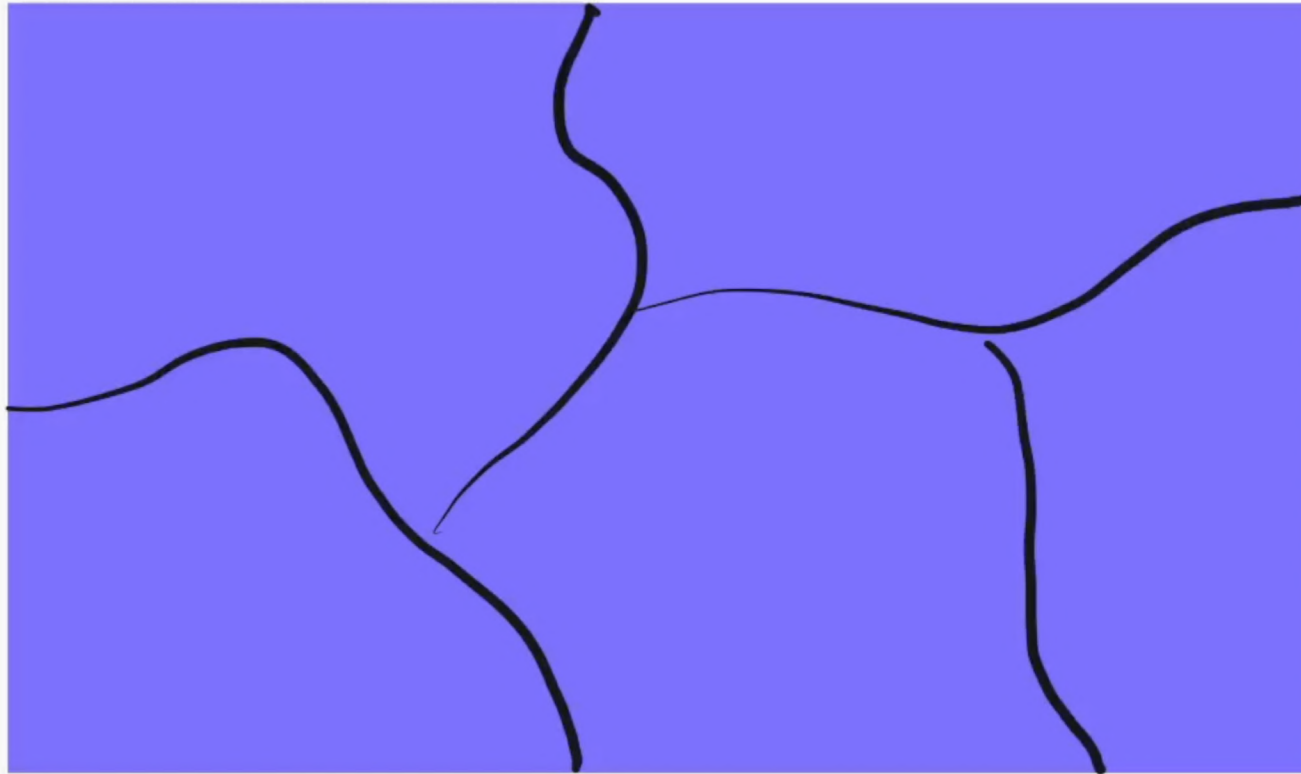
Identification of different microstructural features in a Ti alloy at room temperature.

Phase Transformation during Cooling

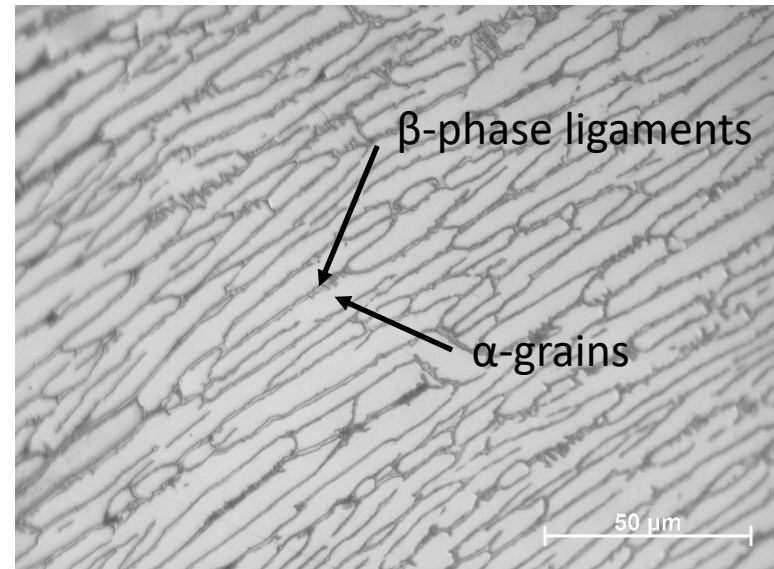
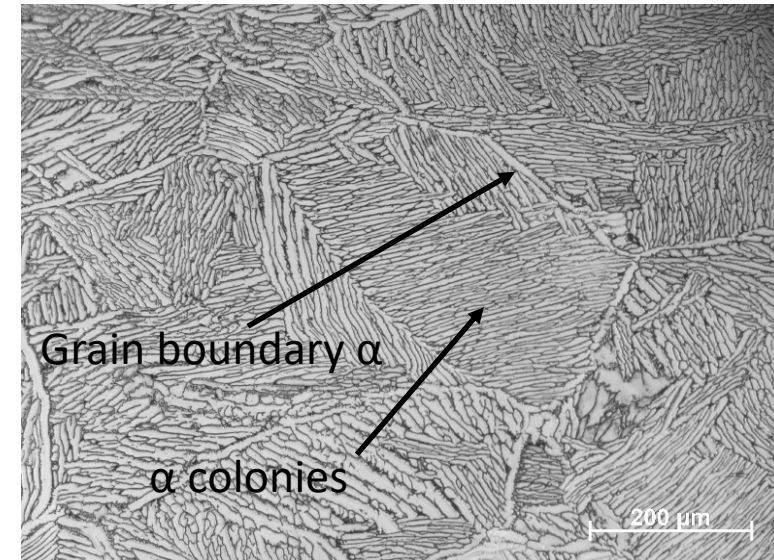
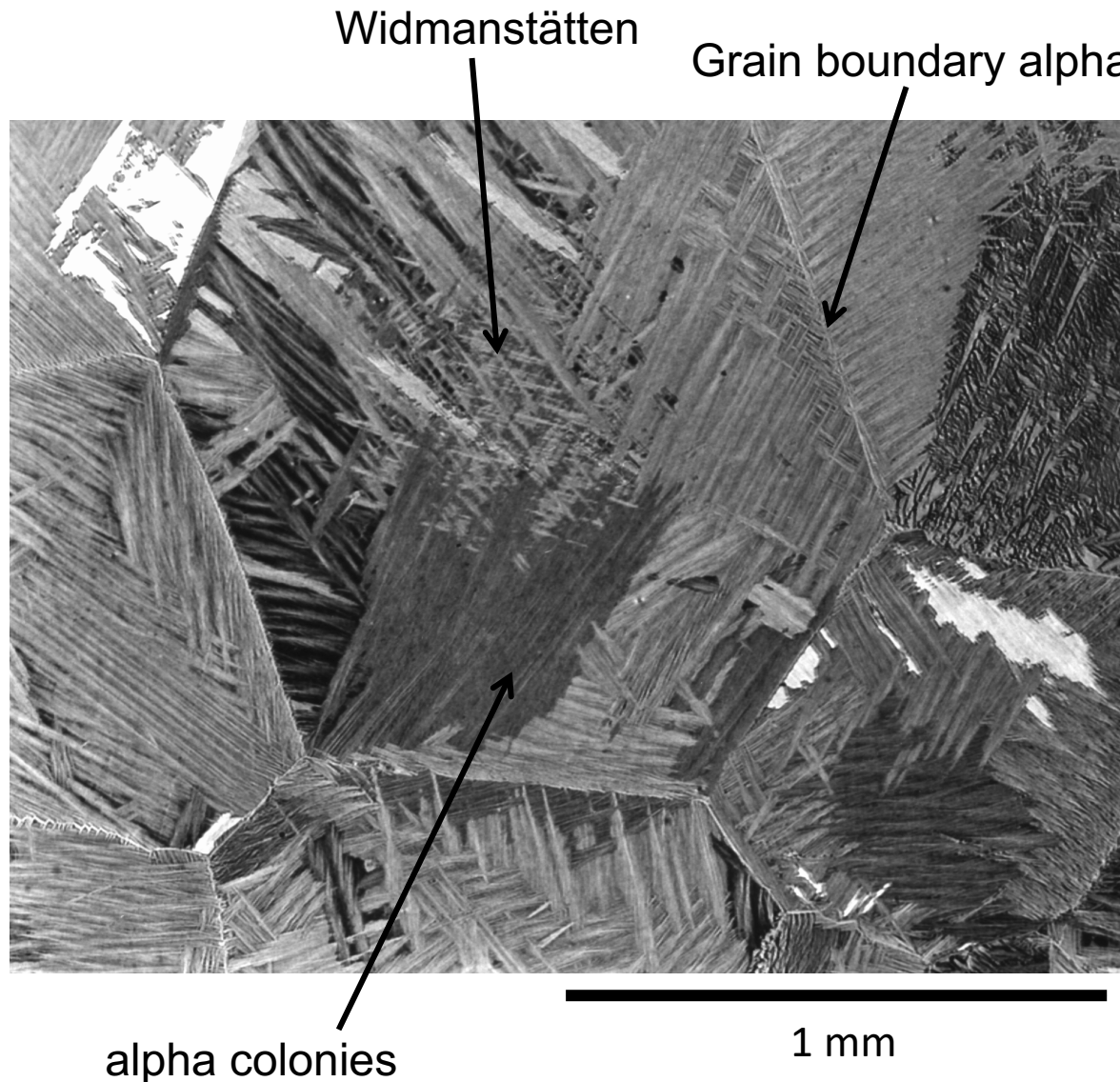
- When casting Ti or Zr alloys the high temperature β -phase forms first (*also important when welding and additive manufacturing*)
- These alloys are single β -phase at very high temperatures ($\gtrsim 900^\circ\text{C}$ in Zr, $\gtrsim 1000^\circ\text{C}$ in Ti).
- α -Ti/Zr tends to form first on the β grain boundaries, before α -lamellae grow into the β -grains.
- α -lath structure forms by epitaxial growth (similar growth of Pearlite in Steel).
- In $\alpha + \beta$ alloys, β -ligaments remain in-between α -laths when cooled to room temperature.



Phase Transformation during Cooling

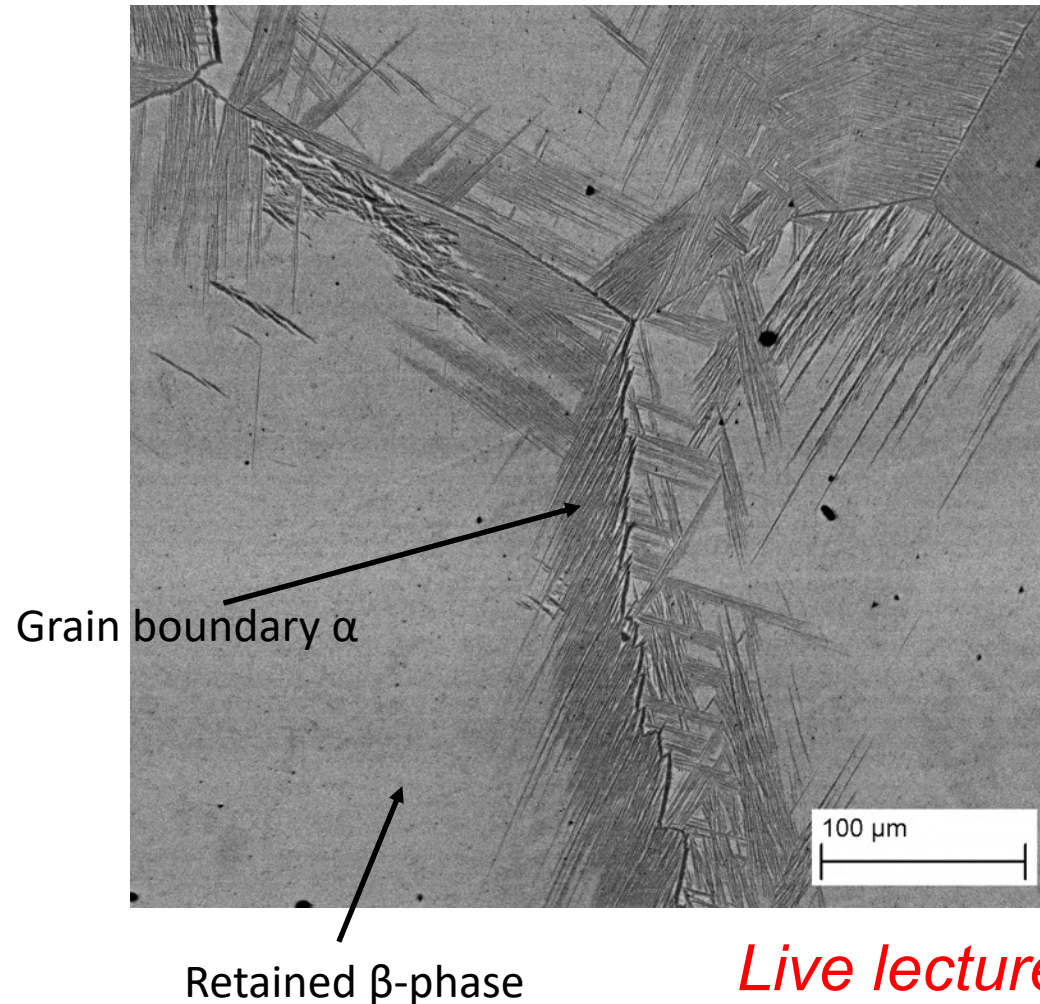


Phase Transformation during Cooling

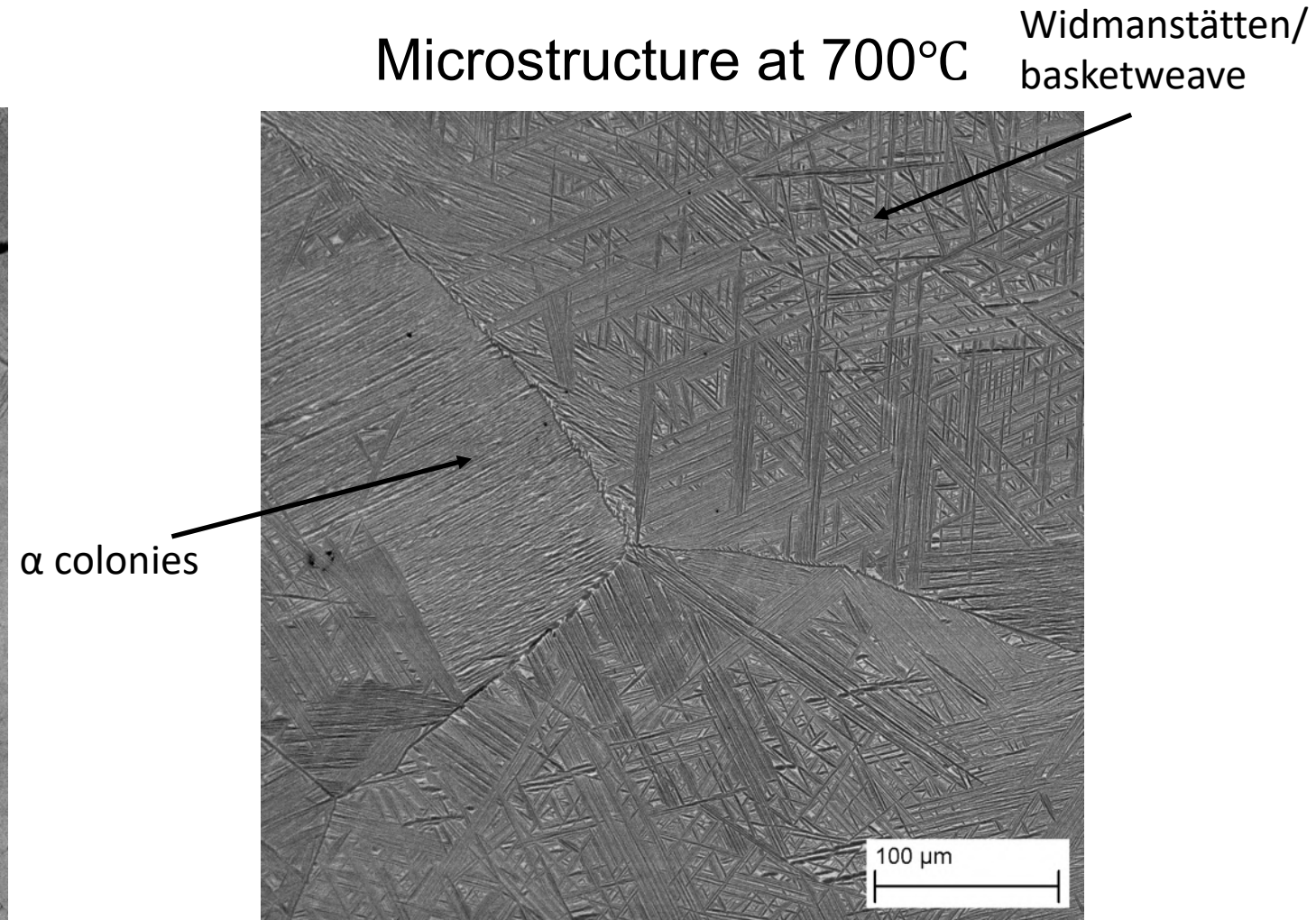


Phase Transformation during Cooling

Microstructure at 800°C



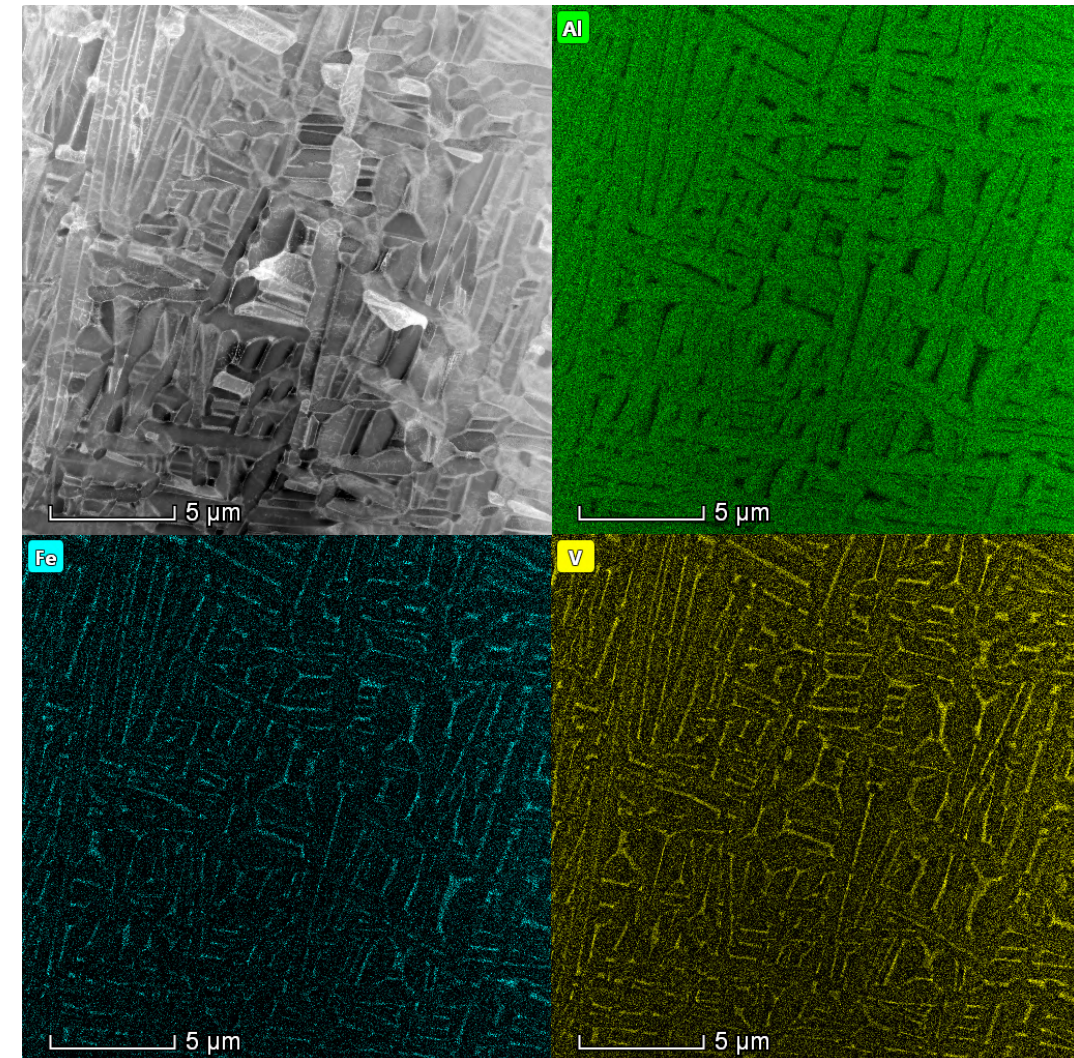
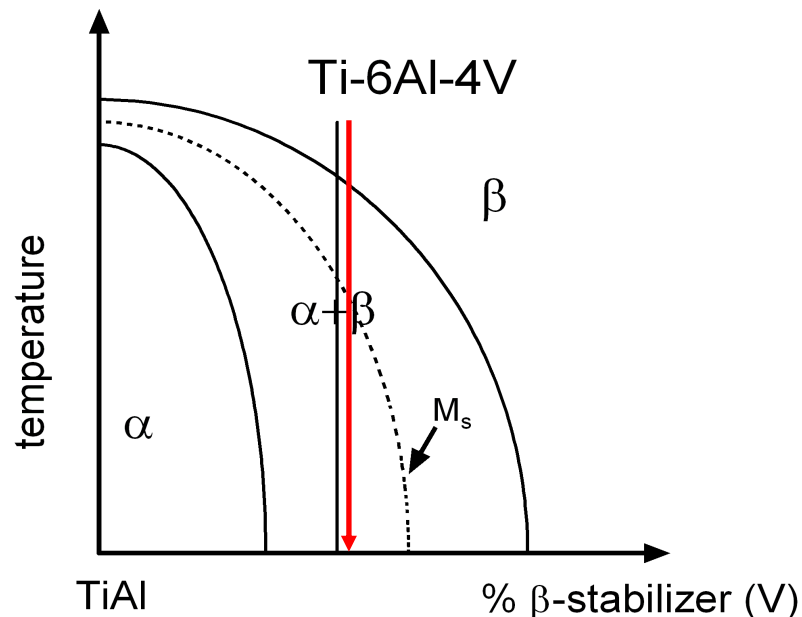
Microstructure at 700°C



Live lecture – Why can we see these microstructures at RT?

Where do the Alloying Additions Go?

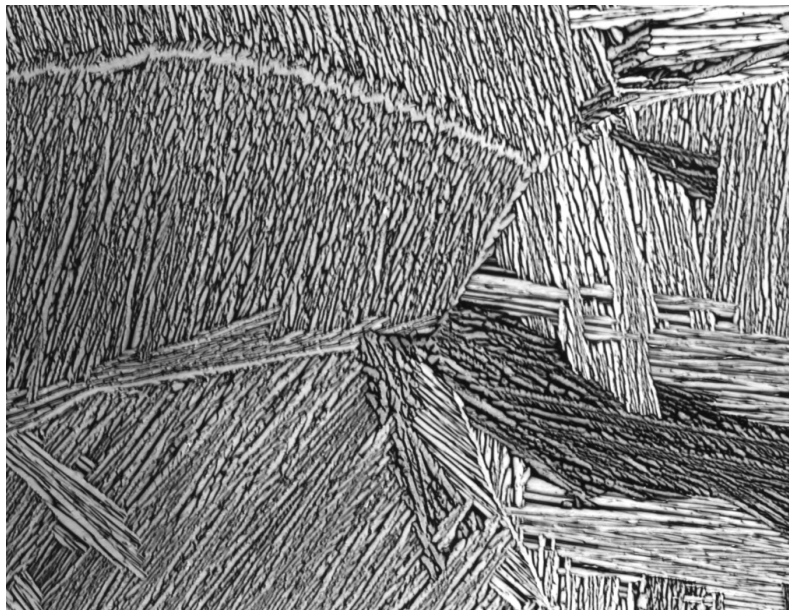
- As temperatures drops, β -phase requires more β -stabilising content to remain stable.
 - β -stabilisers diffuse to β -phase
 - α -stabilisers diffuse to α -phase
- *Thermodynamics determine equilibrium phase fraction.*
 → *Diffusion controls kinetics.*



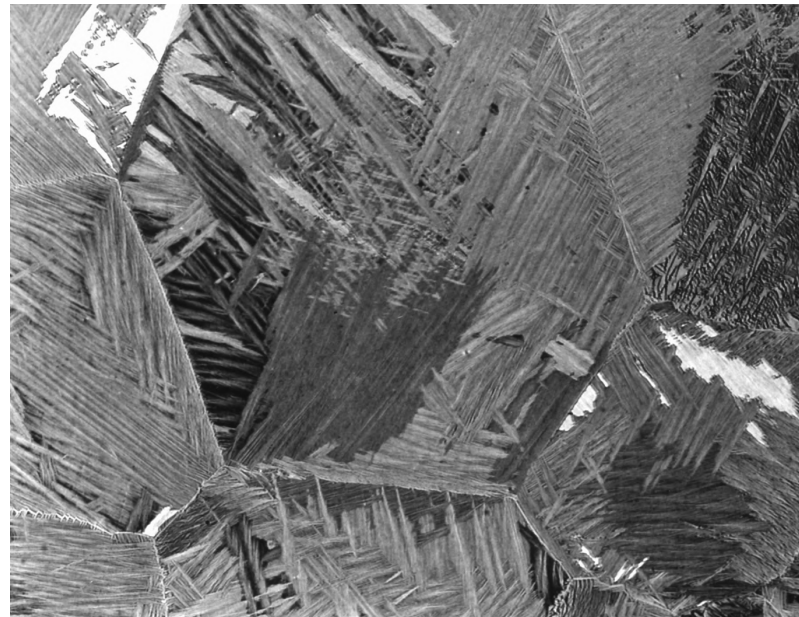
Effect of Cooling Rate

- **Nucleation versus growth;**
 - **At slow cooling rates**, α -Ti (or Zr) tends to form first on the β grain boundaries, before α -lamellae **grow** into the β -grains.
 - **At very fast cooling rates**, fine α -Ti (or Zr) grains can also **nucleate** at sites throughout the β -grains.

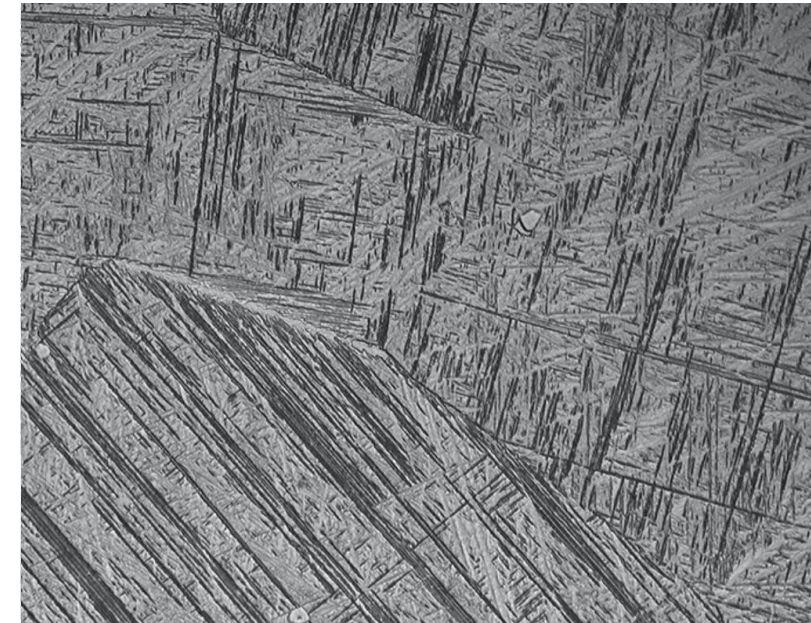
500 microns



slow cooling / furnace cooling
0.1 °C/s



fast cooling / air cooling
1 °C/s



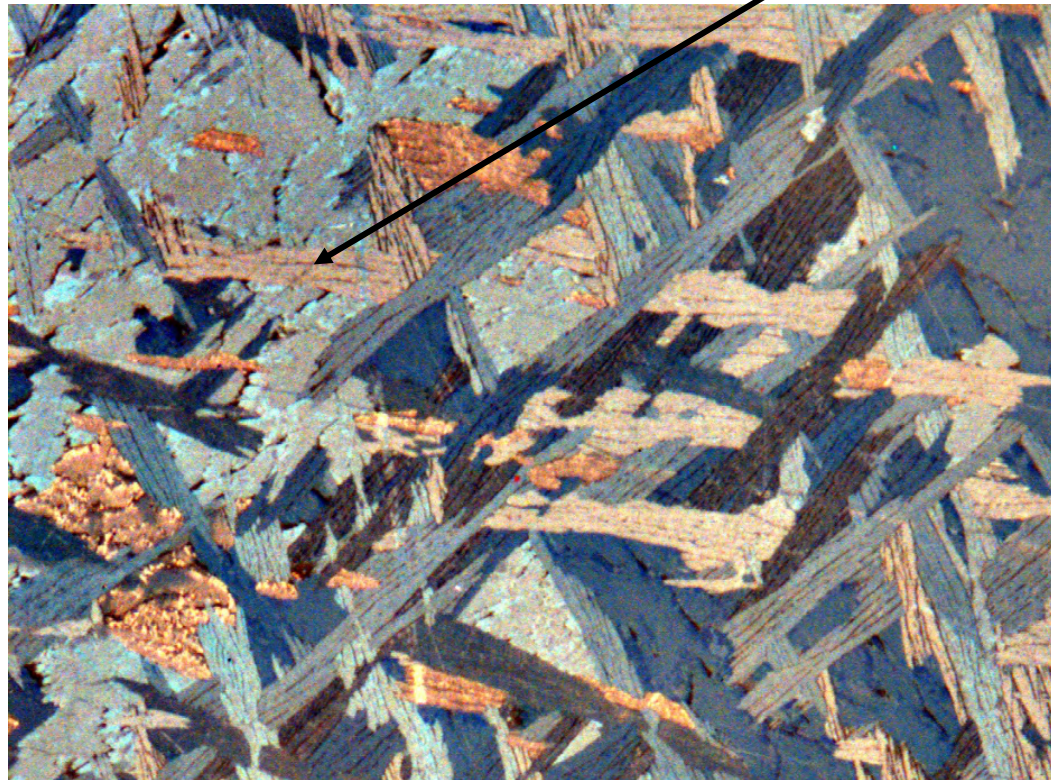
very fast cooling / water quench
> 10 °C/s

Effect of Alloying Addition

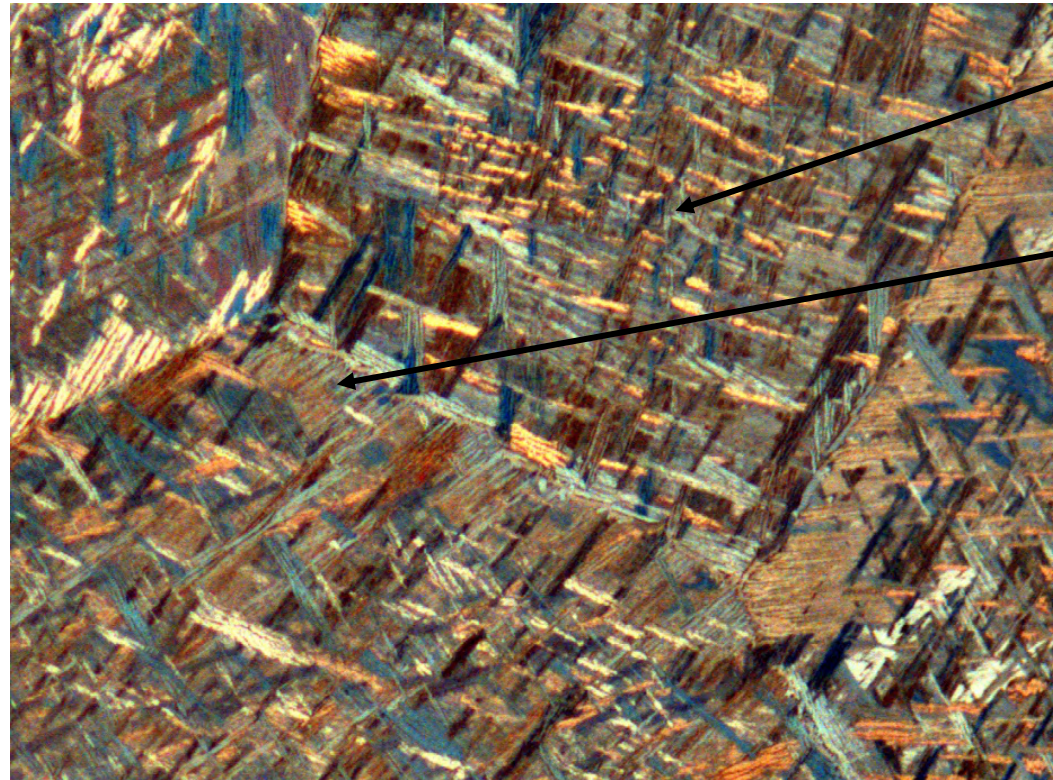
- *Diffusion of alloying elements controls growth (kinetics).*
- *Finer α -laths appear in Zr alloys due to slower diffusing Nb elements (undercooling effect).*

Zr + 2.5 wt.% Nb

Packets of finer α -laths



Zr + 7 wt.% Nb



Widmanstätten/
basketweave

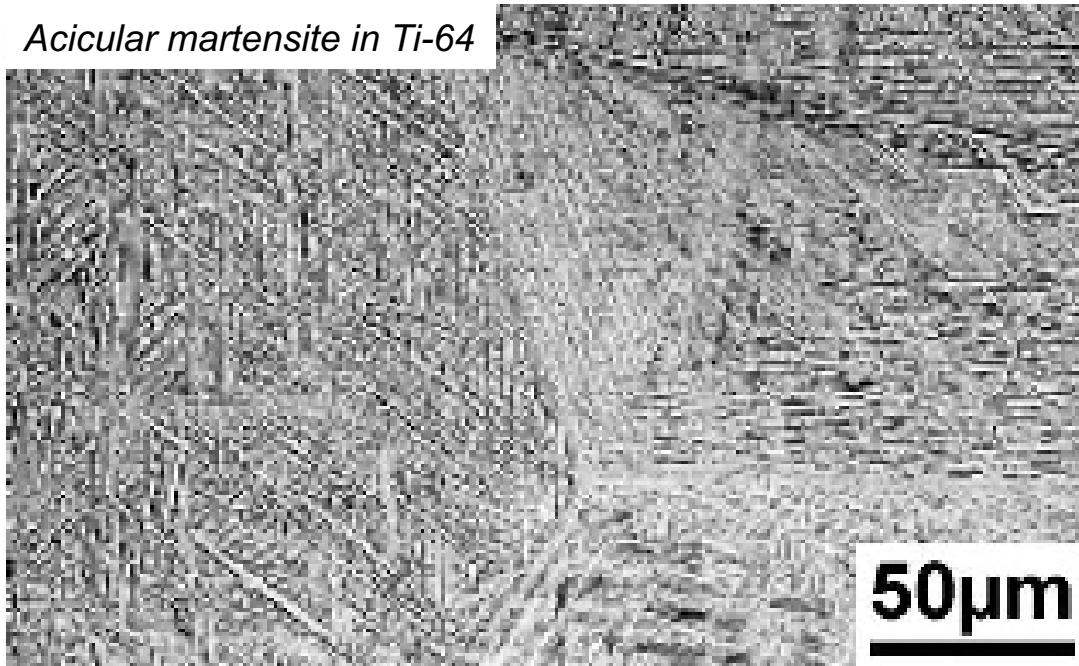
Fine colony α

100 μ m

Allotropic Martensite Transformation (V. Fast Cooling)

- When Ti/Zr is quenched from the beta phase region a martensitic (diffusionless transformation) can occur.
- In contrast to steel, the martensitic transformation in Ti/Zr does not produce a heavily distorted crystal cell.
- Strengthening mechanism in this case mainly results from the grain refinement
 - very fine lamellar (plate) microstructure

Acicular martensite in Ti-64

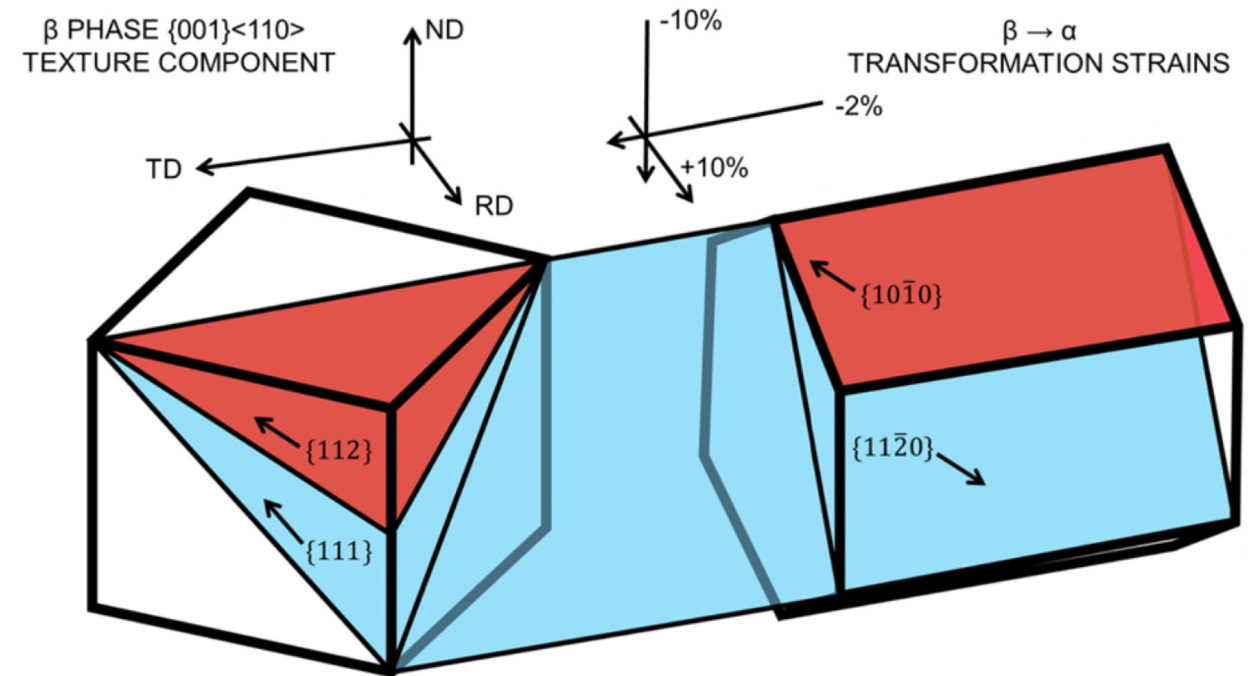


In application, Titanium martensitic microstructures are mainly seen in welds.

Course goal: Describe phase transformation in Ti and Zr alloys during heating and cooling at different rates and explain how this can change the properties of the material.

Learning outcomes:

- Describe the effect of different alloying additions on the phase structure of Ti (and Zr) alloys.
- Describe phase transformation and microstructural changes during heating and cooling of Ti and Zr alloys at different rates.
- ***Recall the Burgers Orientation Relationship (BOR), describing crystallographic relationship between α and β phases, and predict 'variant' orientations.***



Schematic of the Burgers Orientation Relationship (BOR), showing phase transformation in Zr and Ti Alloys.

Burgers Orientation Relationship (BOR)

- Phase transformation governed by Gibbs Free Energy
- ↓
- Small shifts in atomic positions
- ↓
- New crystallographic phase
- **Crystallographic relationship between α and β phase;**

$$\{110\}_{\beta} \parallel (0002)_{\alpha} \text{ and } \langle 111 \rangle_{\beta} \parallel [11\bar{2}0]_{\alpha}$$

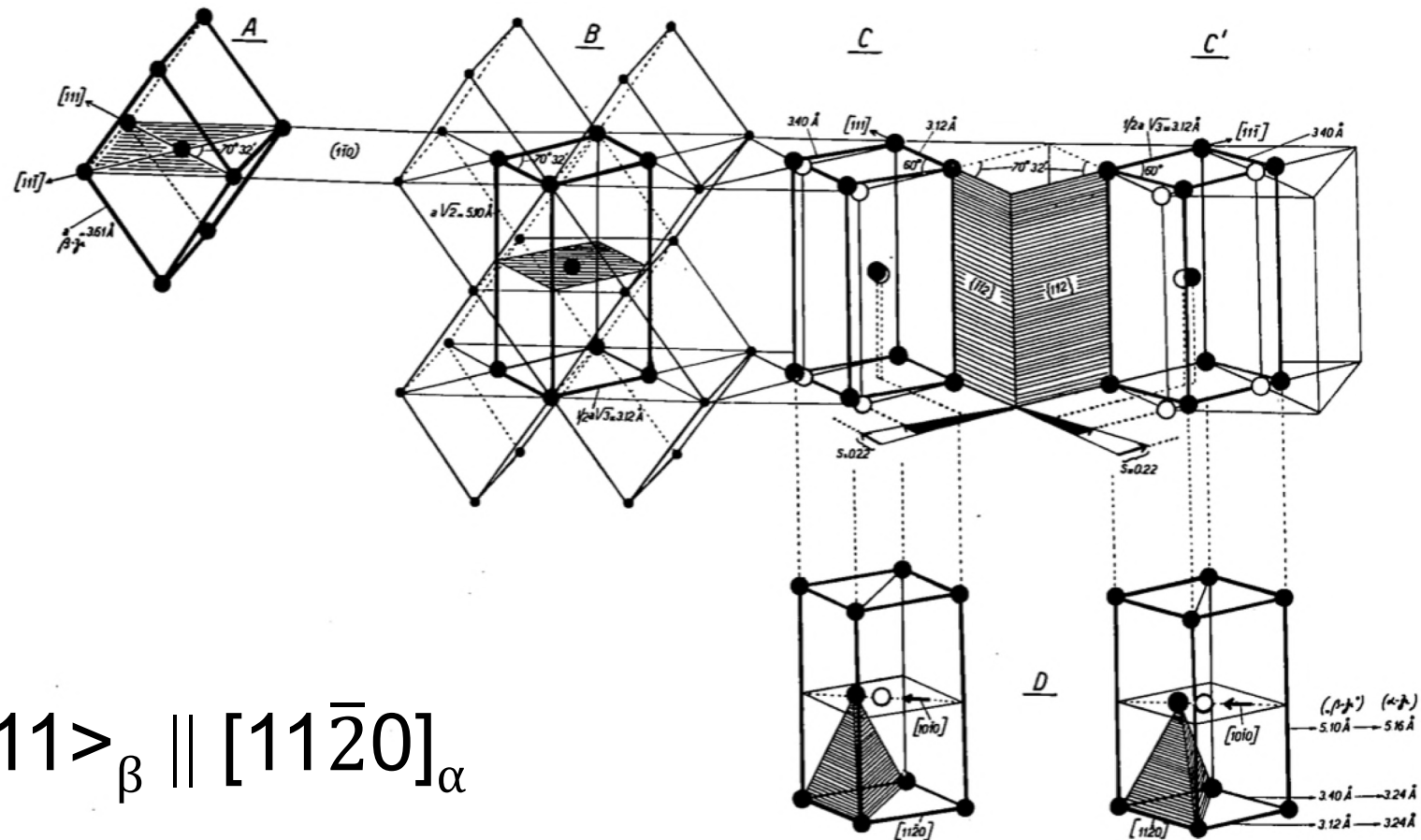
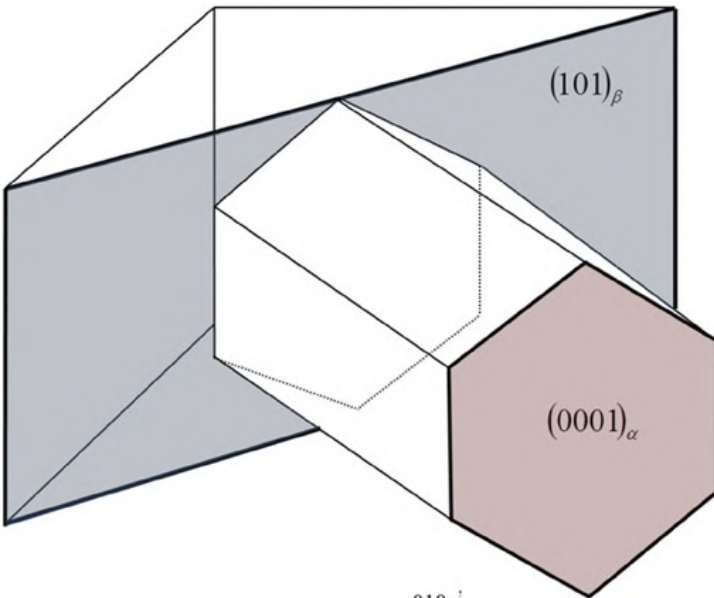


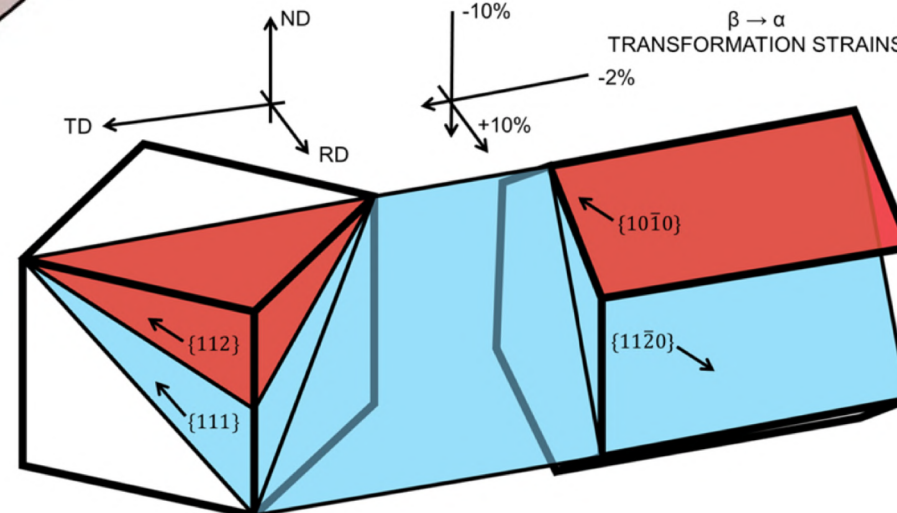
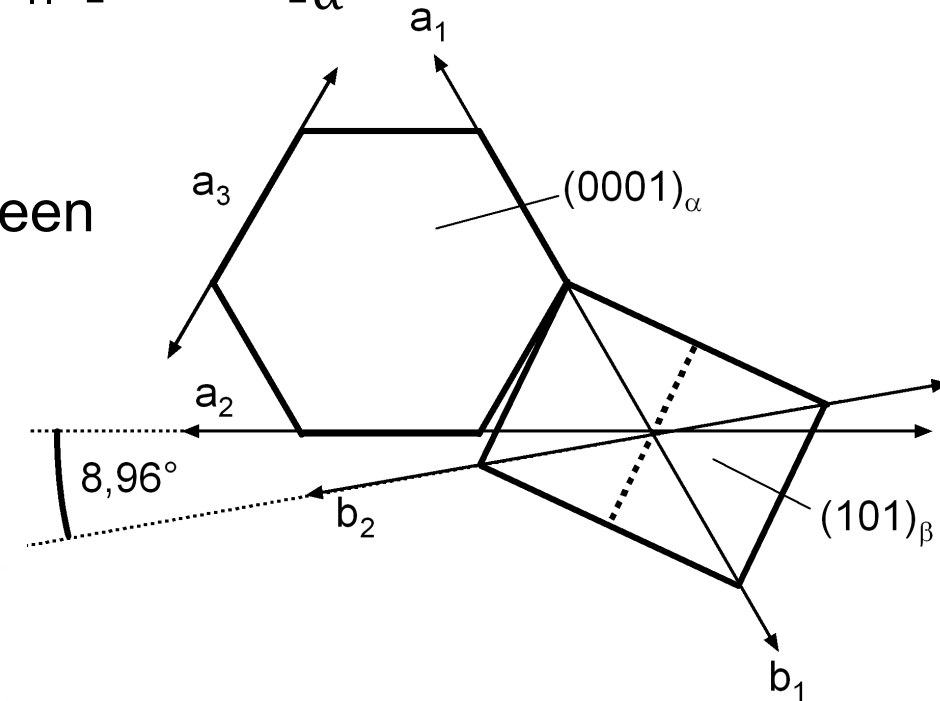
Fig. 5. Transition of the cubic-body-centred into the hexagonal-close-packed lattice in Zirconium-crystals. The figure illustrates how *two* positions of the hexagonal lattice correspond to *one* definite $\{110\}$ -plane.

Burgers Orientation Relationship (BOR)

$$\{110\}_{\beta} \parallel (0002)_{\alpha} \text{ and } \langle 111 \rangle_{\beta} \parallel [11\bar{2}0]_{\alpha}$$



Small angular difference between $\langle 111 \rangle_{\beta} \parallel [1120]_{\alpha}$ alignment

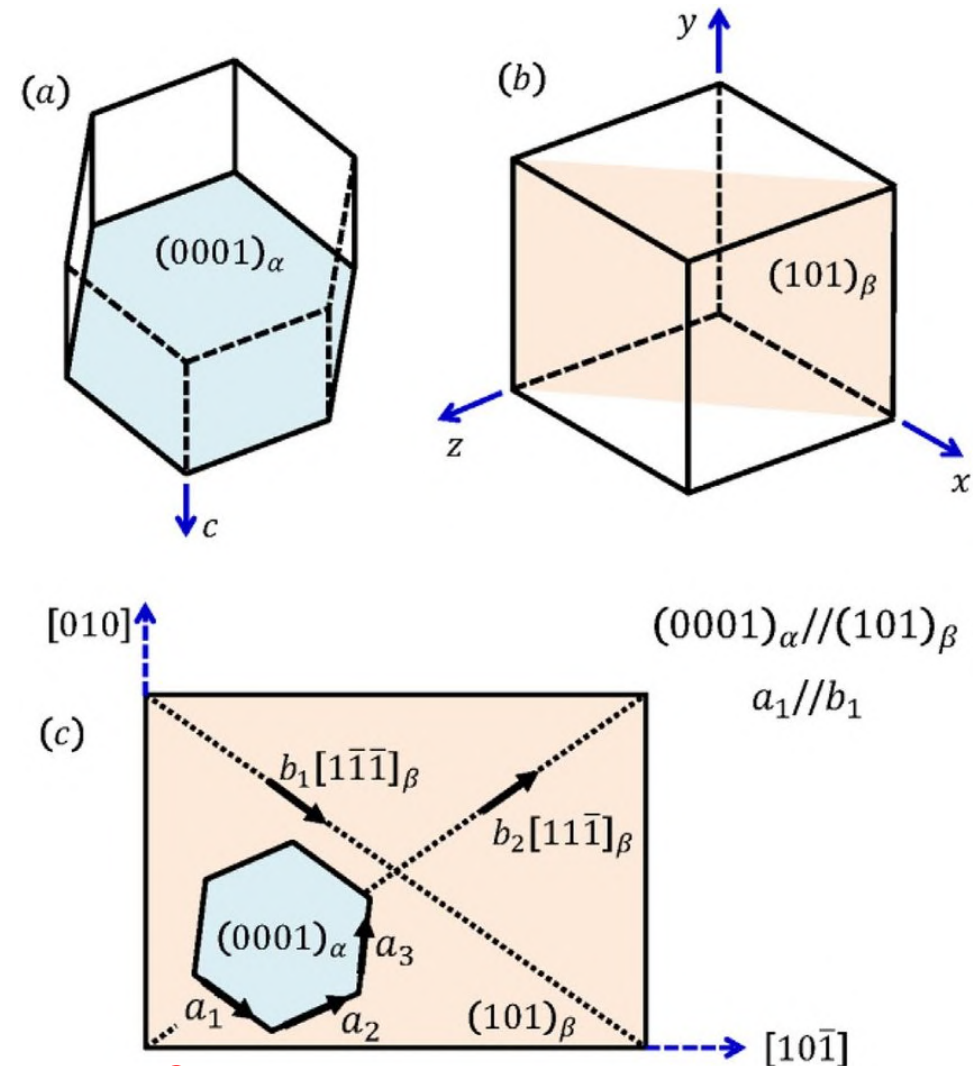


Burgers Orientation Relationship (BOR)

β -phase has;

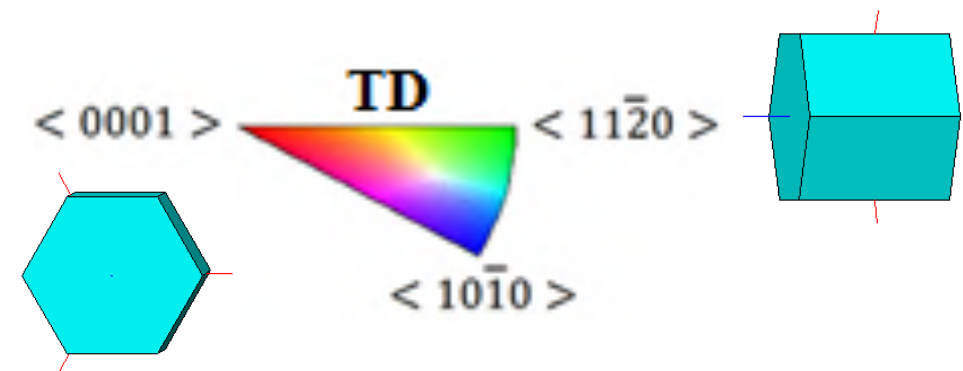
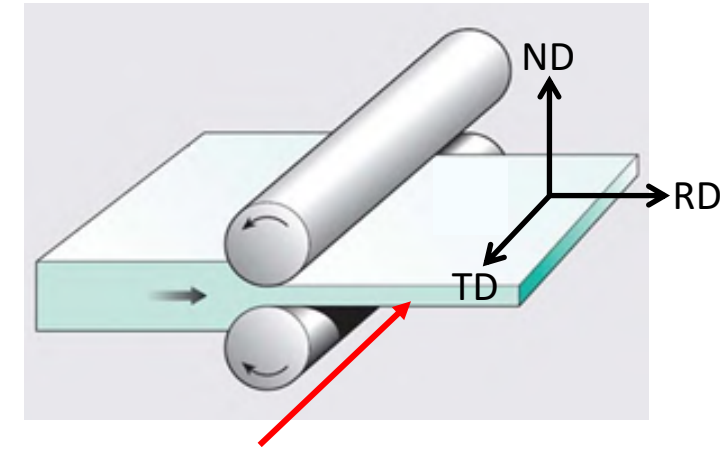
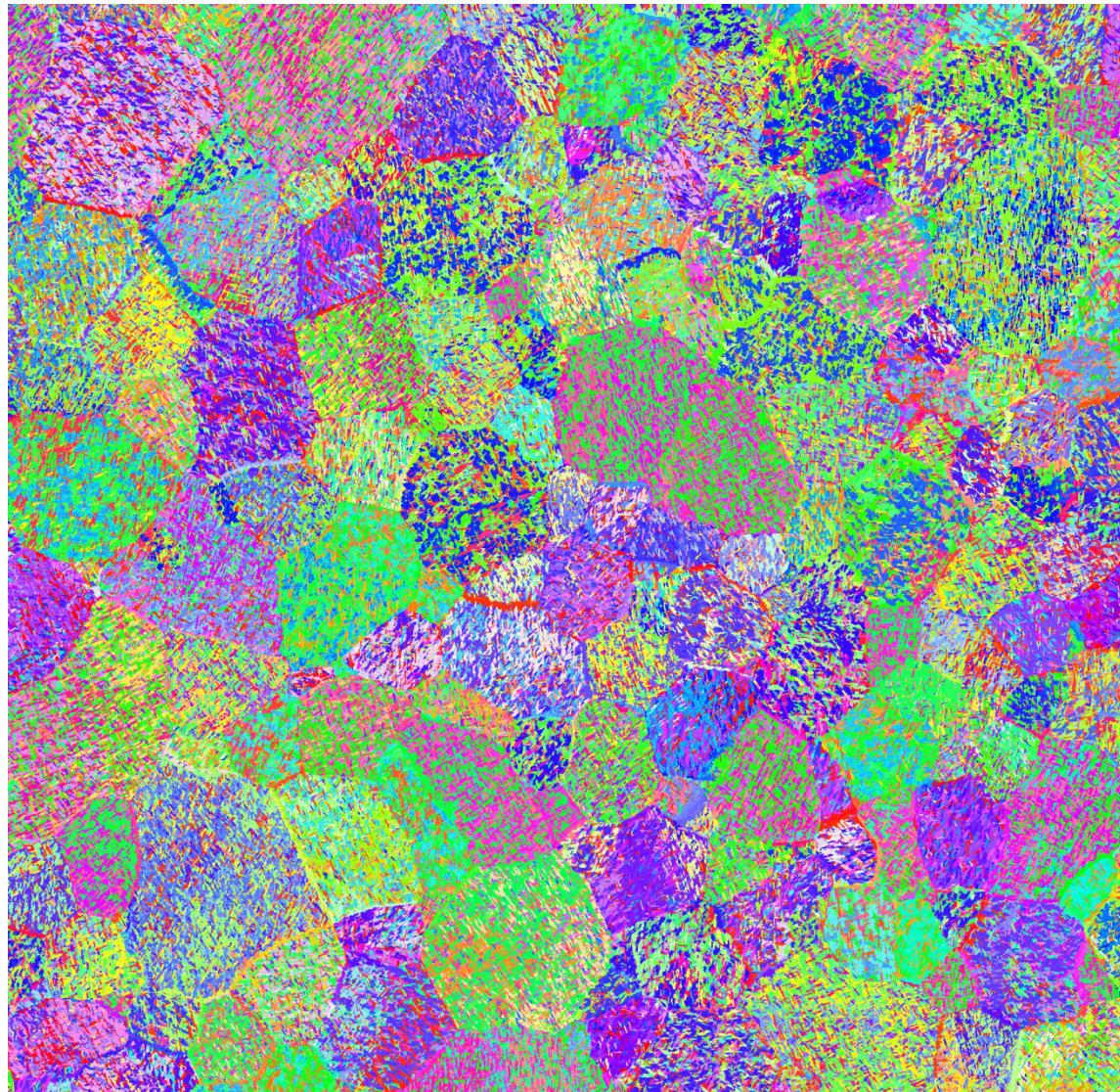
- six different 110 planes
 - each 110 plane contains two 111 directions
- 12 different α variants from a single β grain.

- Note, BOR is maintained during **Heating** and **Cooling**

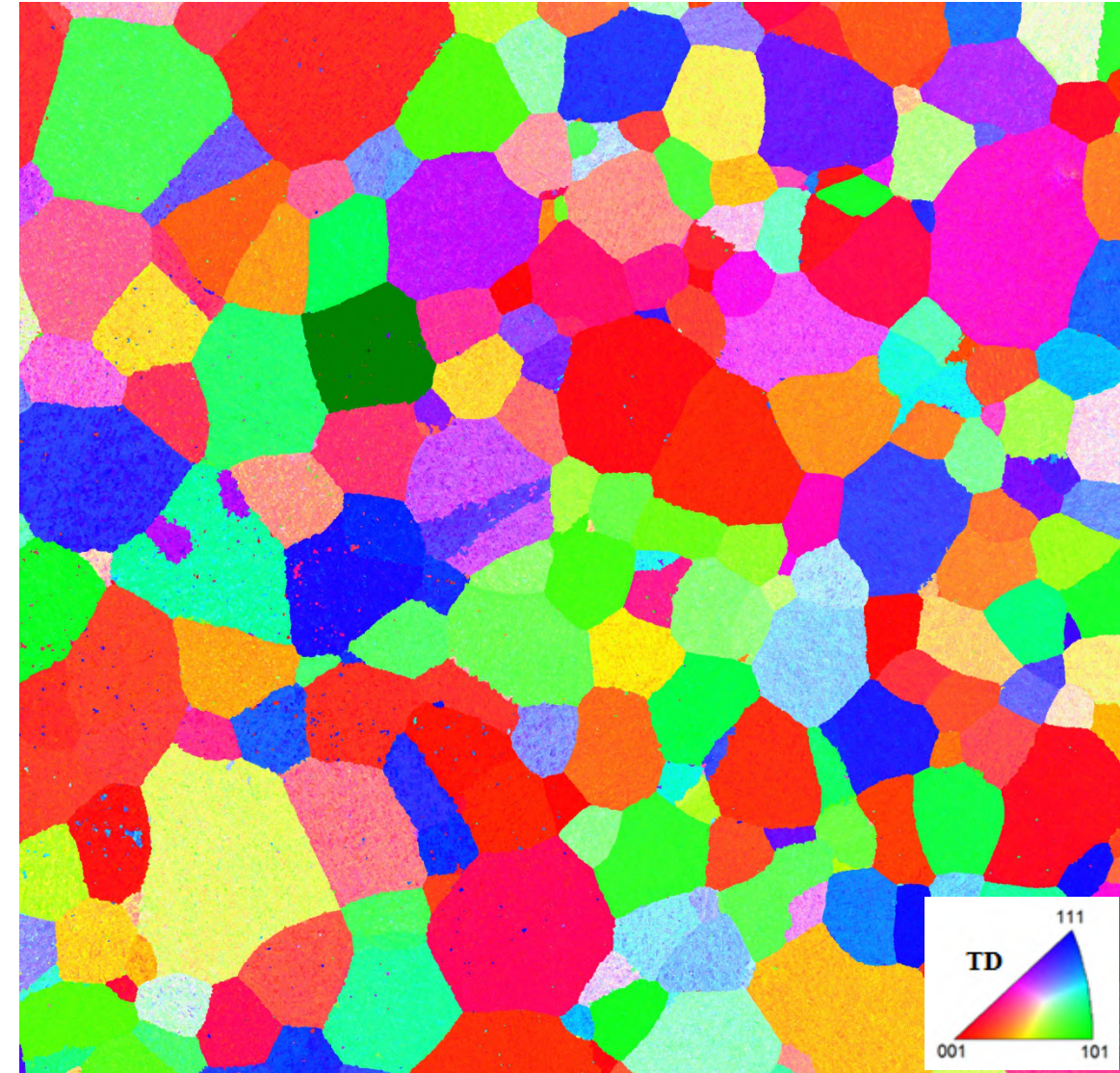
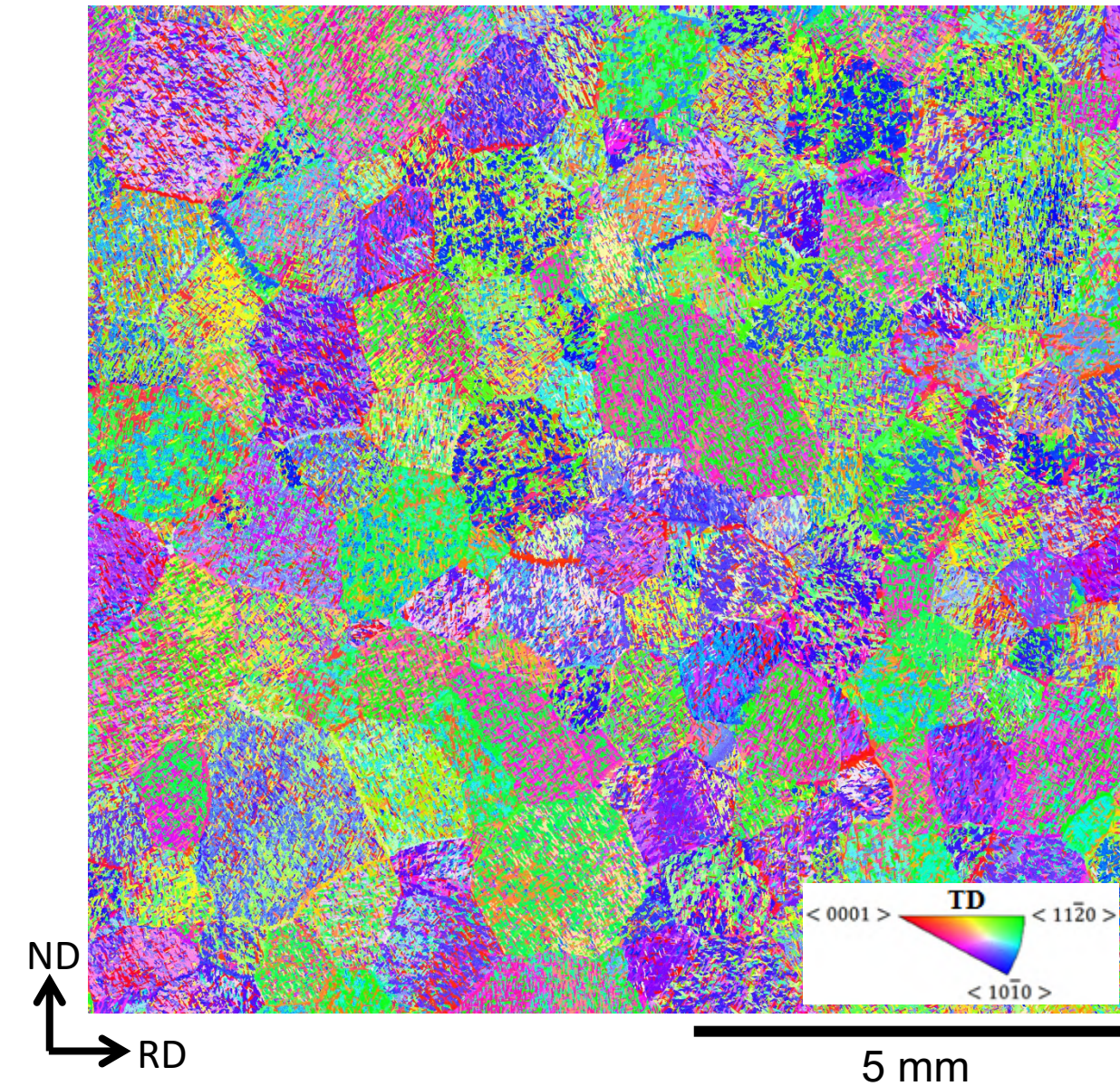


Live lecture – How many possible variants during heating?

Orientation map recorded by Electron Backscatter Diffraction (EBSD)

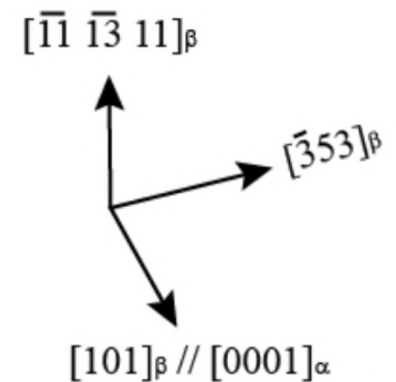
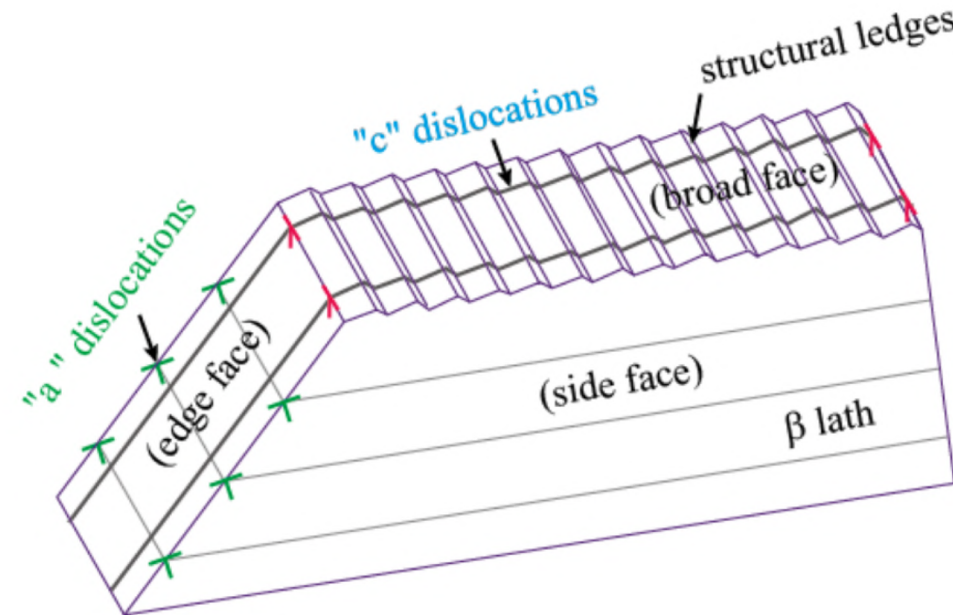
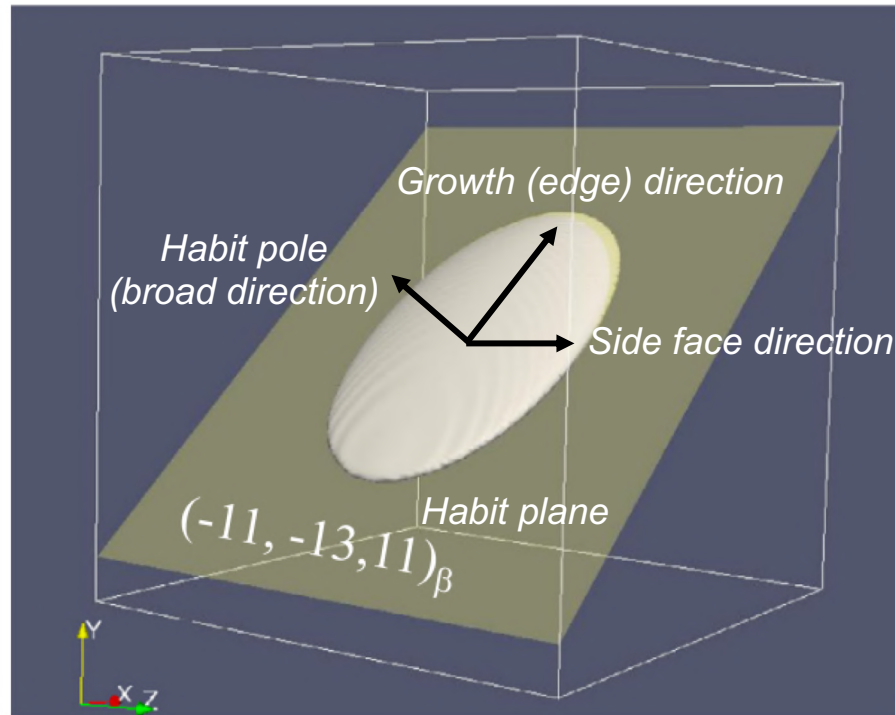


Reconstruction of high temperature β -phase microstructure



Habit (Growth) Plane

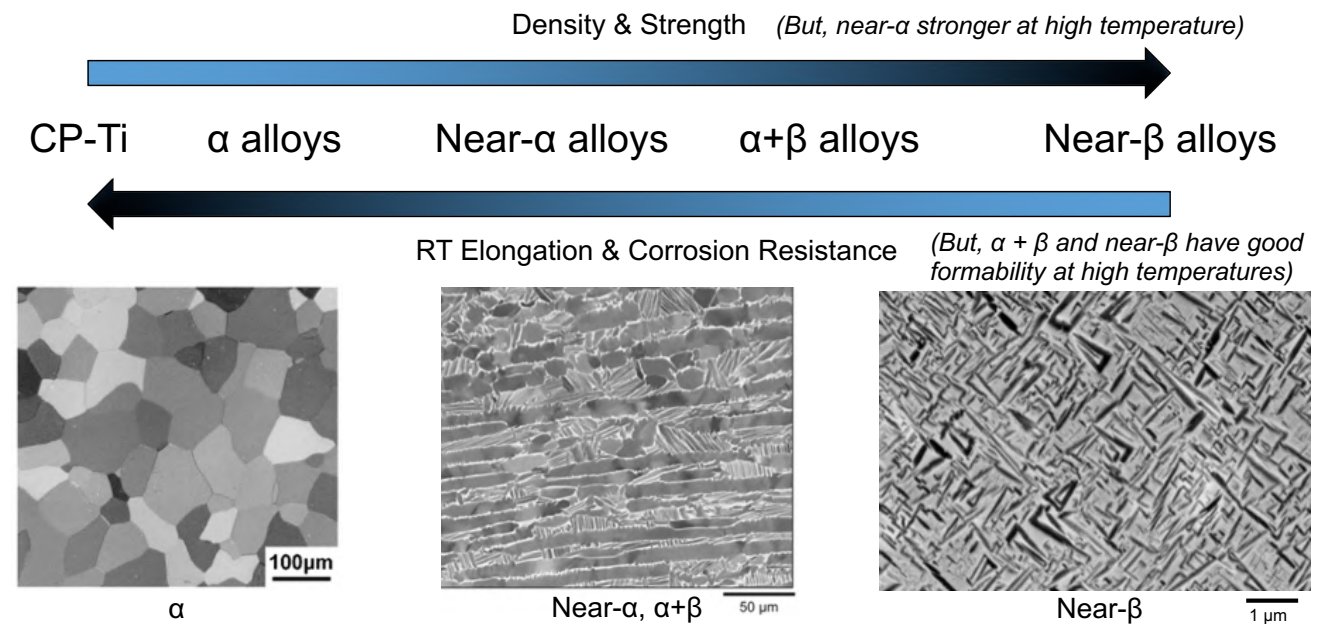
- α -laths are 3D structures. (*But, we often only see them as 2D slices.*)
- Habit plane defines broad face (containing length and breadth directions).
- Growth direction close to $\sim \langle 1\bar{1}1 \rangle_{\beta} || \langle 11\bar{2}0 \rangle_{\alpha}$



Course goal: Describe phase transformation in Ti and Zr alloys during heating and cooling at different rates and explain how this can change the properties of the material.

Learning outcomes:

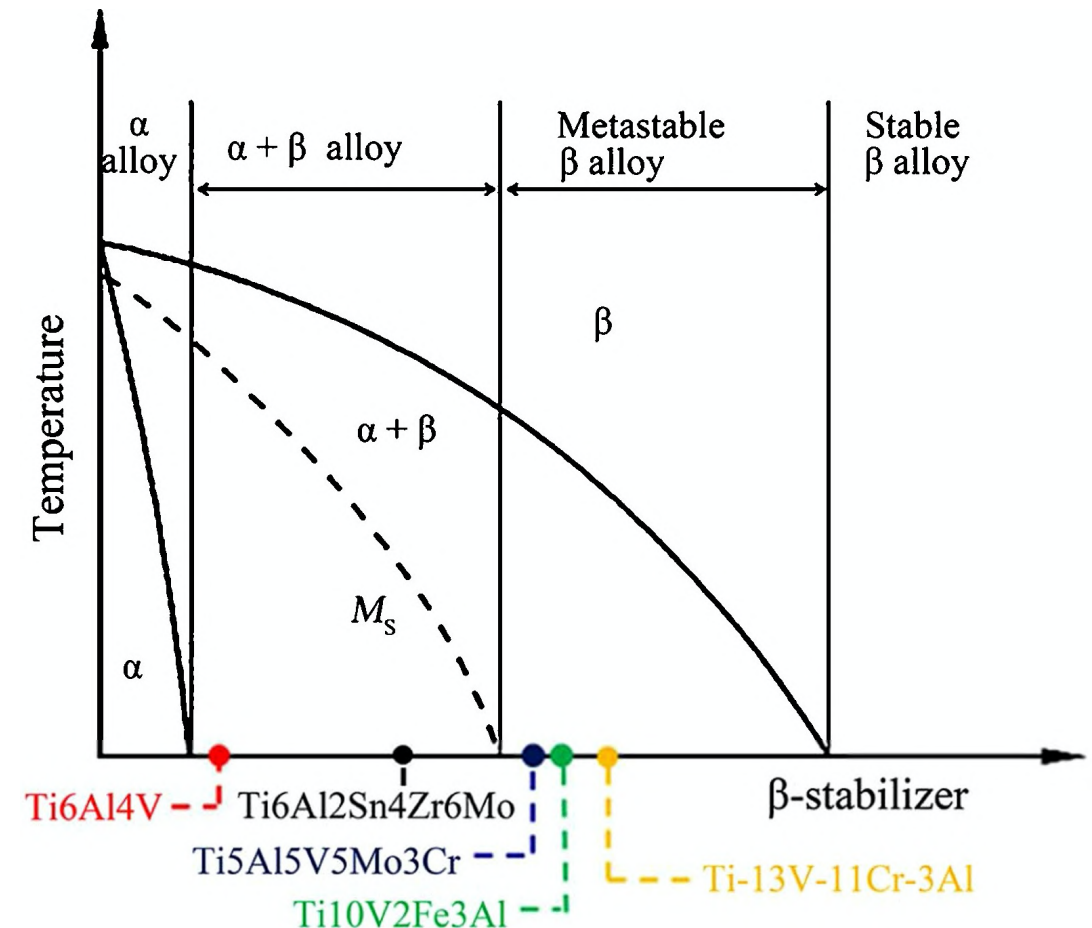
- Describe the effect of different alloying additions on the phase structure of Ti (and Zr) alloys.
- Describe phase transformation and microstructural changes during heating and cooling of Ti and Zr alloys at different rates.
- Recall the Burgers Orientation Relationship (BOR), describing crystallographic relationship between α and β phases, and predict 'variant' orientations.
- Classify different Ti alloys and summarise their different properties and applications.**



General trends in the different classes of Ti alloys.

Classification of Ti Alloys

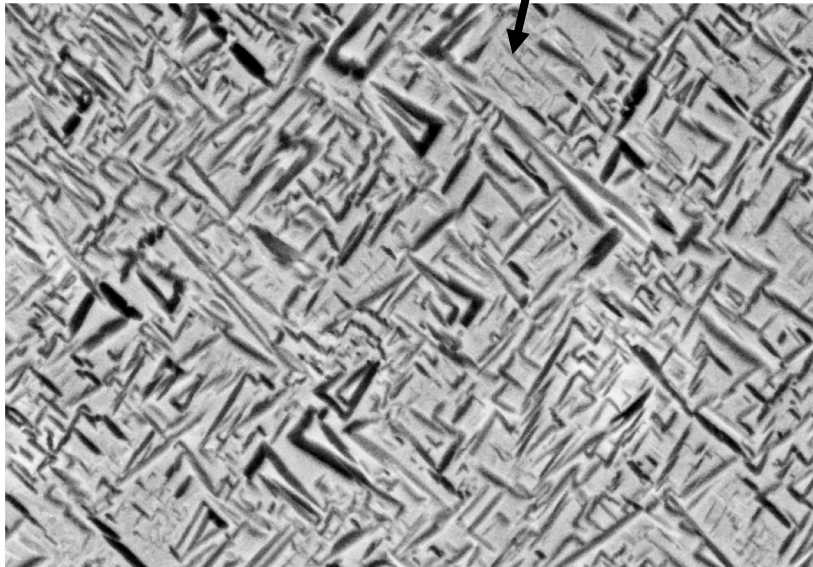
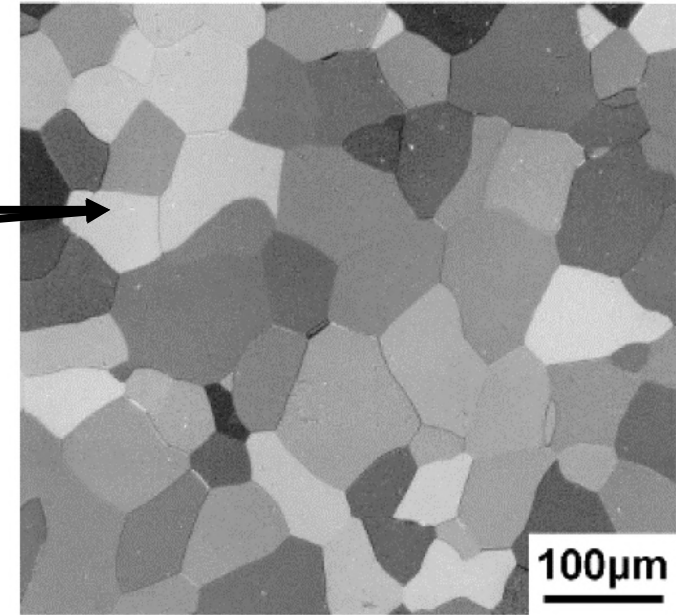
- Commercially Pure Titanium (CP-Ti)
- Single-phase α -Titanium alloys
- Near- α Titanium alloys
- Two-phase $\alpha + \beta$ Titanium alloys
- Near- β (*Metastable*) Titanium alloys



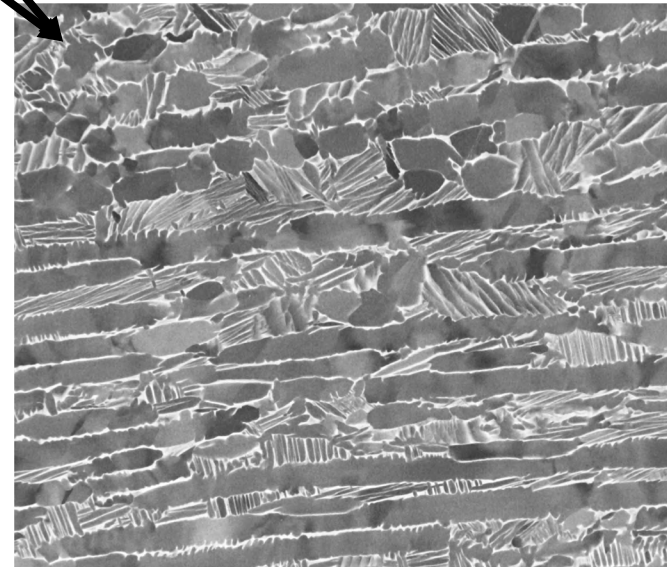
Titanium Classes – Microstructures

α
↓
 $\alpha + \beta$
↓

- commercially pure (CP) titanium
- α alloys
- near- α alloys
- $\alpha + \beta$ alloys
- near- β alloys



1 μm

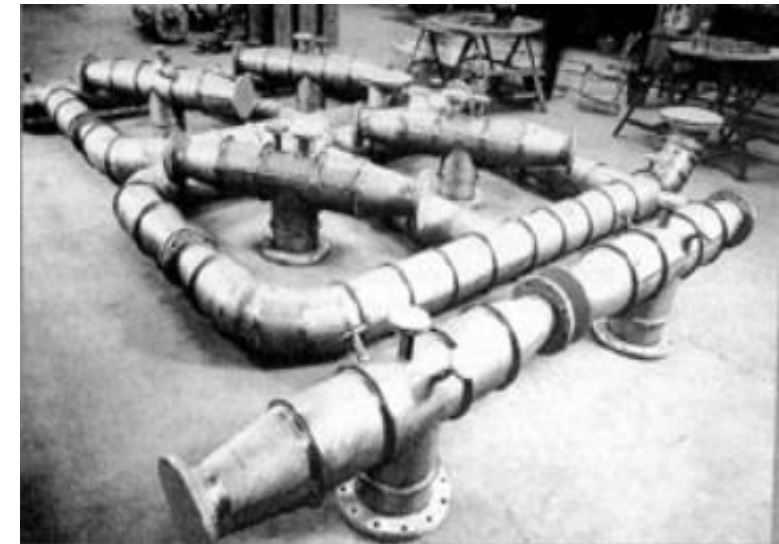


50 μm

Note the difference in
scales!

Commercially Pure Titanium (CP-Ti)

- **Applications:** Chemical and Petrochemical industries as piping, heat exchangers, pumps and valves.
- **Properties:** Superior corrosion resistance, high formability, good weldability. Low strength.
- *CP-Ti (Grade 1 and 2) can be cold-rolled.*



Pipework for offshore use made from CP-Ti

Grade or Alloy	O (max.)	Fe (max.)	$\sigma_{0.2}$ (MPa)
----------------	----------	-----------	-------------------------

CP Titanium

CP Titanium Grade 1	0.18	0.20	170
CP Titanium Grade 2	0.25	0.30	275
CP Titanium Grade 3	0.35	0.30	380
CP Titanium Grade 4	0.40	0.50	480

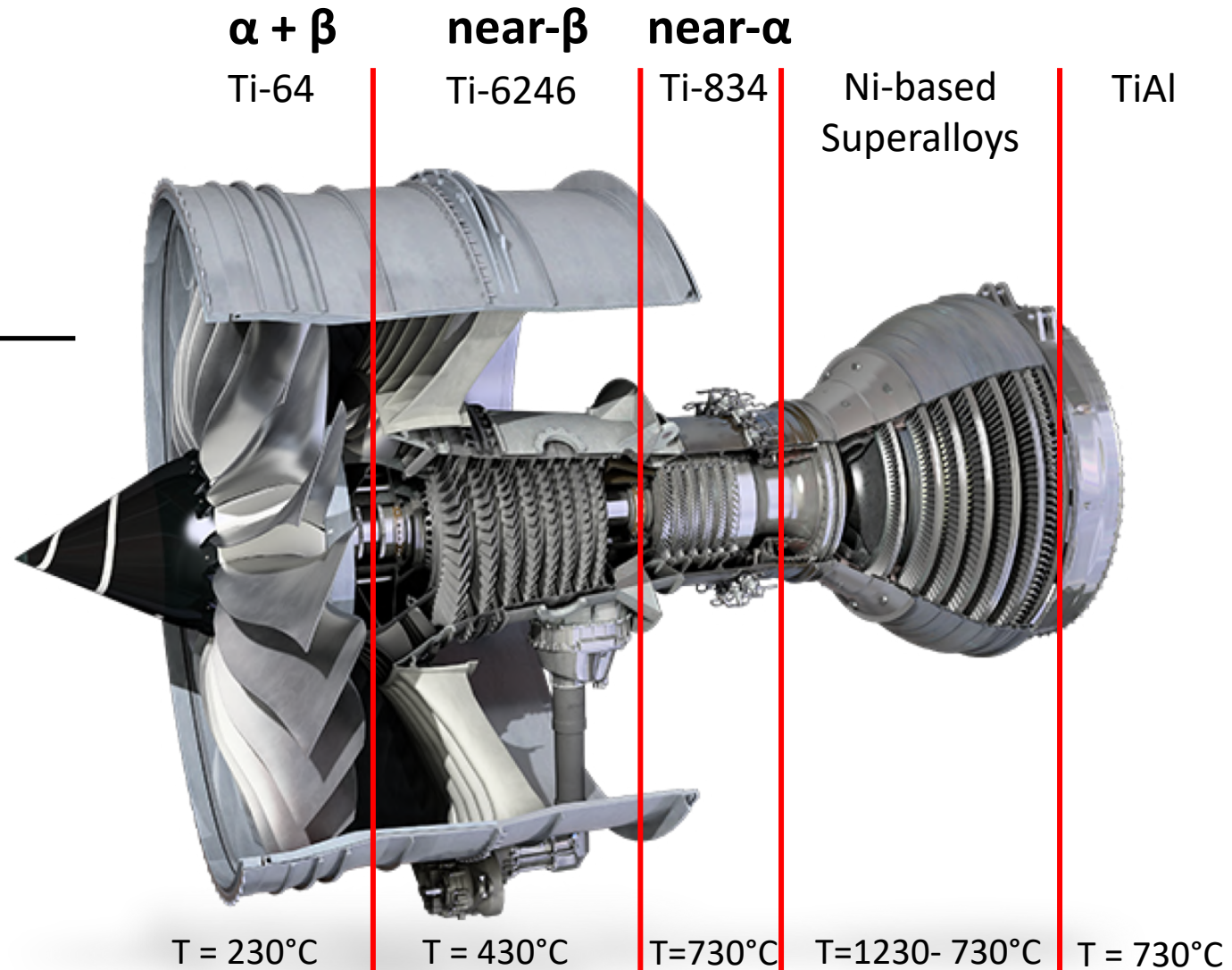
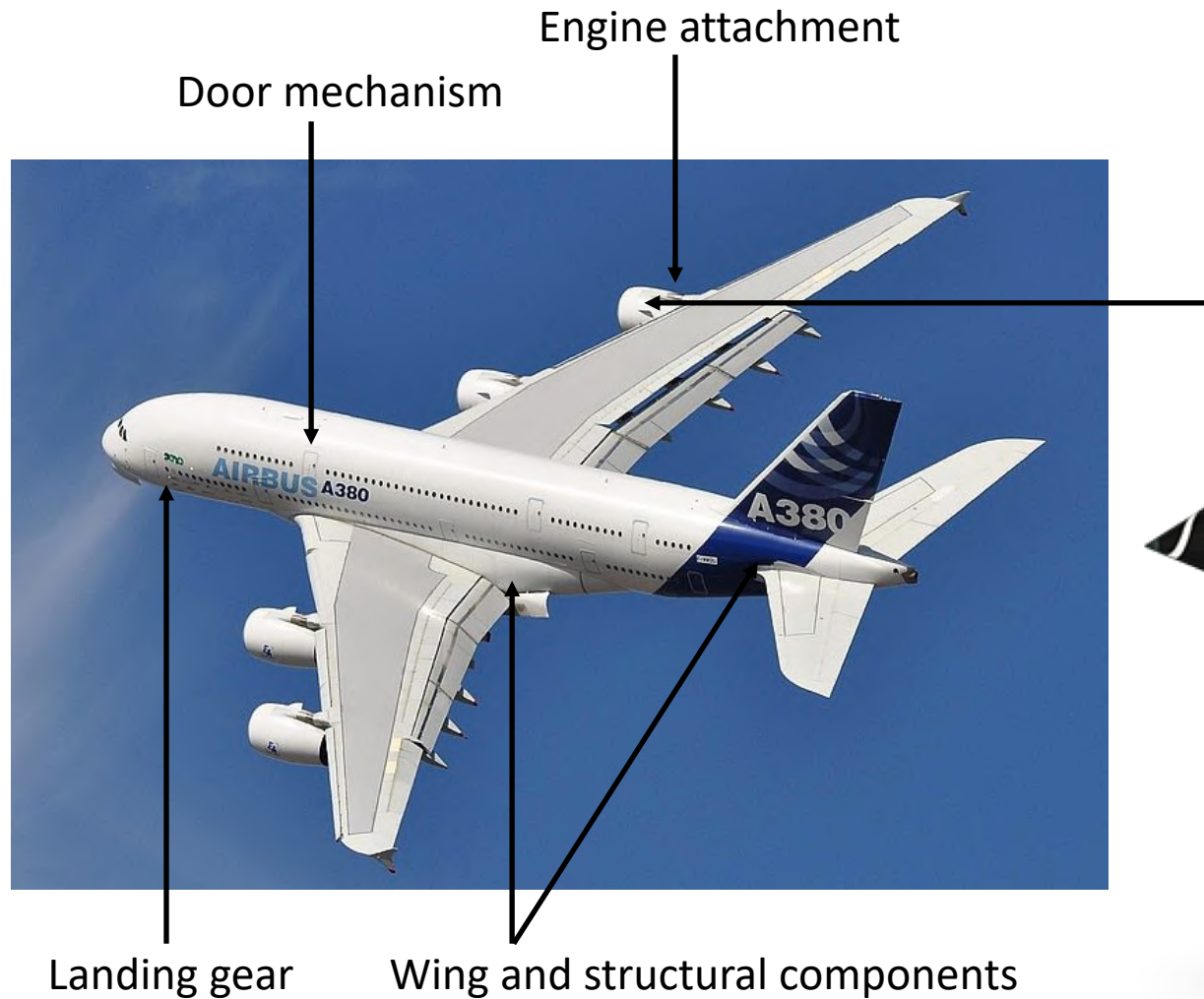
*Reduced
elongation (%)
and ductility*

Single-phase α -Titanium alloys

- **Applications:** Cryogenic applications (with low level of interstitials). Otherwise, not very common as difficult to process.
- **Properties:** Good strength, toughness, creep properties and weldability. Good corrosion behaviour. Poor formability at low temperatures.
- *Strengthened via alloying additions, and thermomechanical processing.*

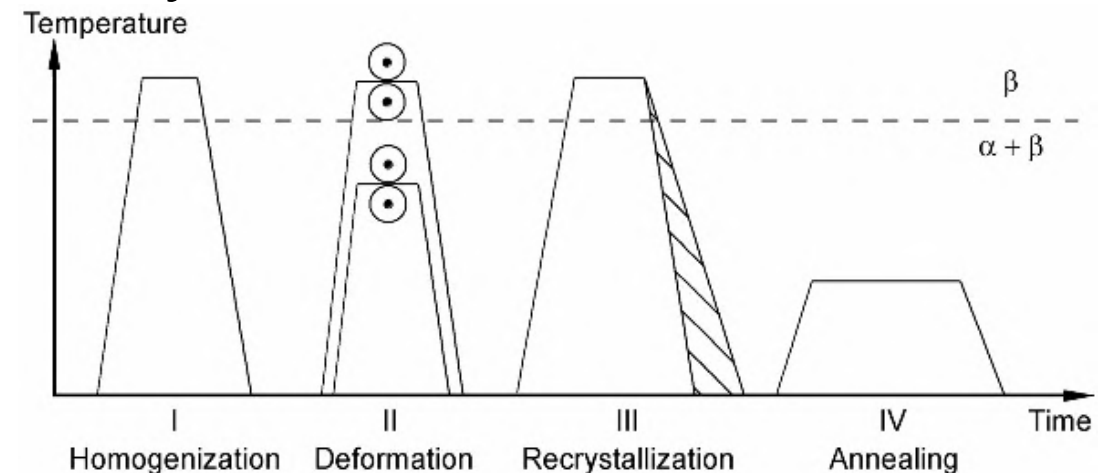
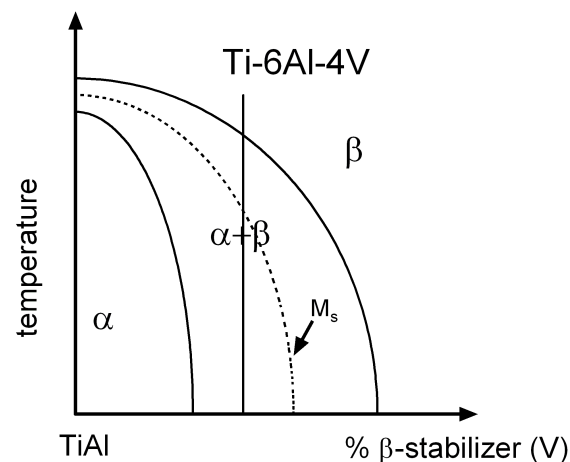
Grade or Alloy	O (max.)	Fe (max.)	Other Additions	$\sigma_{0.2}$ (MPa)	<div><div>Solid Solution Strengthening</div><div>Reduced elongation (%) and ductility</div></div>
α Titanium Alloys					
Ti-0.3Mo-0.9Ni (Grade 12)	0.25	0.30	0.2-0.4Mo, 0.6-0.9Ni	345	
Ti-3Al-2.5V (Grade 9)	0.15	0.25	2.5-3.5Al, 2.0-3.0V	485	
Ti-3Al-2.5V-0.05Pd (Grade 18)	0.15	0.25	2.5-3.5Al, 2.0-3.0V, (+Pd)	485	
Ti-3Al-2.5V-0.1Ru (Grade 28)	0.15	0.25	2.5-3.5Al, 2.0-3.0V, (+Ru)	485	
Ti-5Al-2.5Sn (Grade 6)	0.20	0.50	4.0-6.0Al, 2.0-3.0Sn	795	
Ti-5Al-2.5Sn ELI	0.15	0.25	4.75-5.75Al, 2.0-3.0Sn	725	

Ti Alloys in the Aerospace Industry



Two-phase $\alpha + \beta$ Titanium alloys

- **Applications:** Aerospace industry (blades, compressor discs, fasteners). Offshore industry (heat exchanger). Elevated temperature applications (up to 400°C)
- **Properties:** Higher strength (stronger than CP-Ti and α -Ti). Ok corrosion resistance. Poorer creep behaviour. Properties highly dependent on microstructure!
- *Thermomechanical processed at high temperatures, with sufficient amount of β -phase.*
- *Ti-6Al-4V (Ti-64) is the work horse of all Ti alloys.*

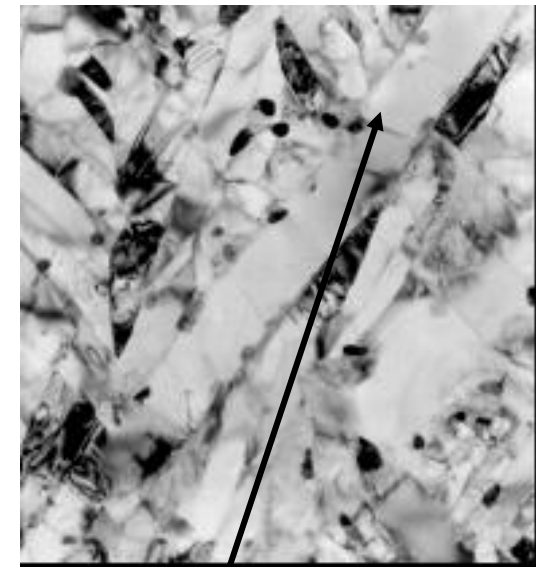


Near- α Titanium alloys

- **Applications:** Medium compressor parts in jet engine (aerospace industry)
- **Properties:** High strength (comparable to Ti-64) and improved creep behaviour. Capable of operating at higher temperatures (up to 700°C). Difficult to process, machine and weld (than Ti-64).
- *Less β -stabiliser (than Ti-64) \rightarrow increased strength and volume of more creep-resistant α -phase at high temp. \rightarrow improved high temperature capability.*
- *Si Additions: contribute to solid solution strengthening and formation of silicides (to improve creep properties)*



High pressure compressor disk made of Timetal-834



silicides

250nm

Important Near- α and $\alpha + \beta$ and Ti alloys

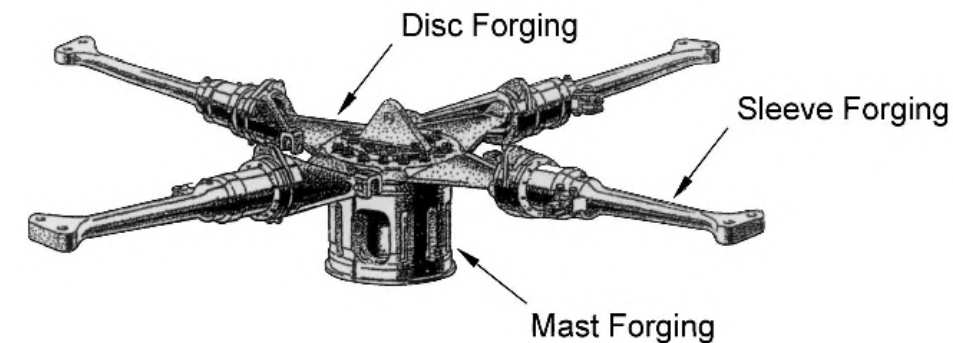
Common Name	Alloy Composition (wt%)	T_{β} (°C)
$\alpha + \beta$ Alloys		
Ti-811	Ti-8Al-1V-1Mo	1040
IMI 685	Ti-6Al-5Zr-0.5Mo-0.25Si	1020
IMI 834	Ti-5.8Al-4Sn-3.5Zr-0.5Mo-0.7Nb-0.35Si-0.06C	1045
Ti-6242	Ti-6Al-2Sn-4Zr-2Mo-0.1Si	995
Ti-6-4	Ti-6Al-4V (0.20O)	995
Ti-6-4 ELI	Ti-6Al-4V (0.13O)	975
Ti-662	Ti-6Al-6V-2Sn	945
IMI 550	Ti-4Al-2Sn-4Mo-0.5Si	975

Near- β Ti alloys

- **Applications:** Larger compressor parts, landing gear, sheets, fasteners, helicopter rotor.
- **Properties:** Good weldability and formability. Hardened to very high strength. Improved corrosion resistance (better than Ti-64, worse than α -Ti). High hydrogen tolerance.
- *Can retain 100% β -phase when quenched.*
- *Excellent formability in solution treated condition.*



A380 landing gear



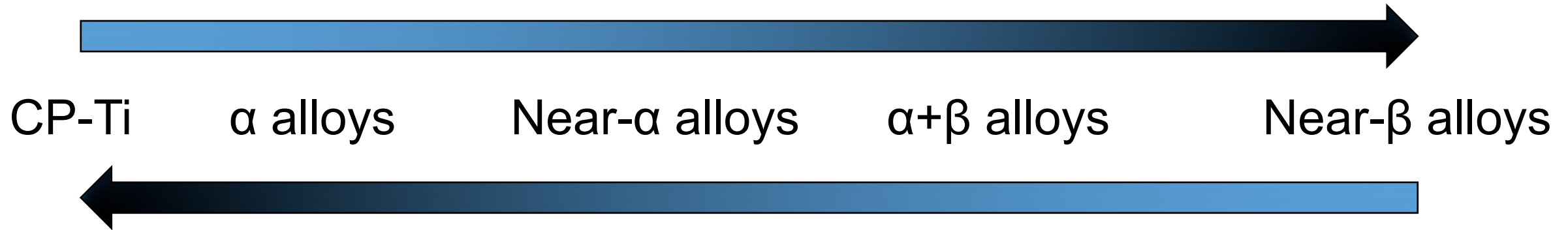
Helicopter rotor

Important Near- β Ti alloys

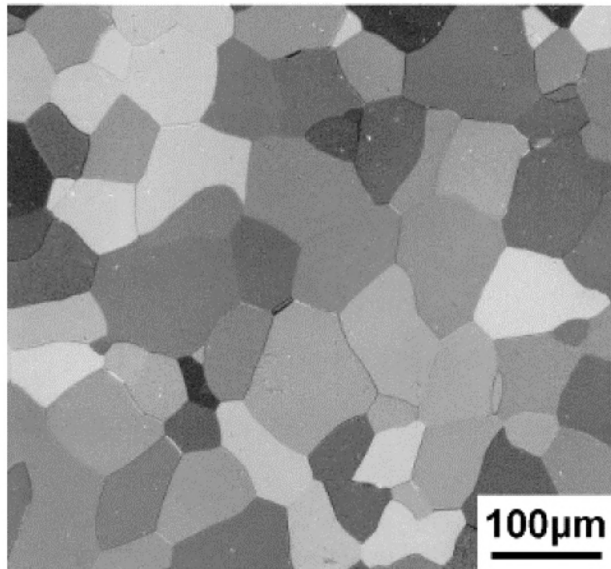
Common Name	Alloy Composition (wt%)	T_{β} (°C)	
β Alloys			
Ti-6246	Ti-6Al-2Sn-4Zr-6Mo	940	<div> Lower β-transus with more β-stabilising addition. </div>
Ti-17	Ti-5Al-2Sn-2Zr-4Mo-4Cr	890	
SP-700	Ti-4.5Al-3V-2Mo-2Fe	900	
Beta-CEZ	Ti-5Al-2Sn-2Cr-4Mo-4Zr-1Fe	890	
Ti-10-2-3	Ti-10V-2Fe-3Al	800	
Beta 21S	Ti-15Mo-2.7Nb-3Al-0.2Si	810	
Ti-LCB	Ti-4.5Fe-6.8Mo-1.5Al	810	
Ti-15-3	Ti-15V-3Cr-3Al-3Sn	760	
Beta C	Ti-3Al-8V-6Cr-4Mo-4Zr	730	
B120VCA	Ti-13V-11Cr-3Al	700	

Titanium Alloy Class General Trends

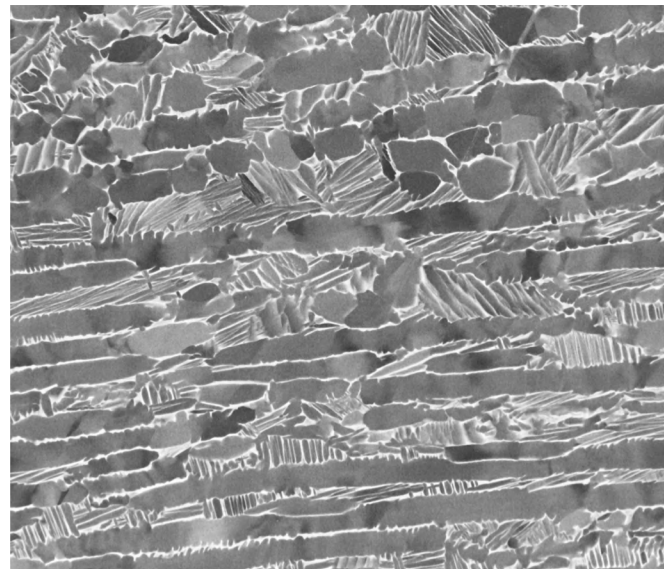
Density & Strength *(But, near- α stronger at high temperature)*



RT Elongation & Corrosion Resistance *(But, $\alpha + \beta$ and near- β have good formability at high temperatures)*

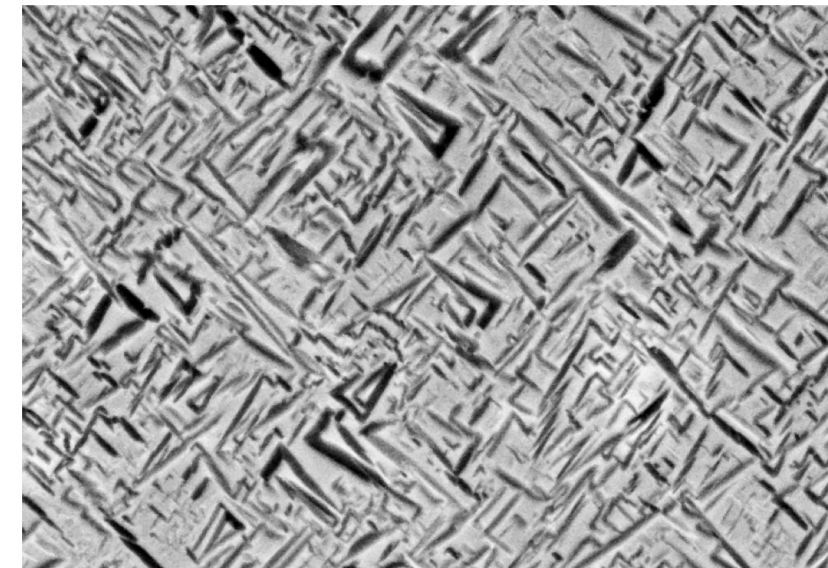


α



Near- α , $\alpha+\beta$

50 μm



Near- β

1 μm